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Setting Bar-Bending Requirements for High-Strength Steel Bars

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Setting Bar-Bending Requirements for High-Strength Steel Bars

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Dedication

This thesis is dedicated to my parents, Hong Yan Dai and Rong Zhao, for their endless love, support and encouragement. They have always been my source of strength, optimism and inspiration in life.

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Abstract

Setting Bar-Bending Requirements for High-Strength Steel Bars

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The University of Texas at Austin, 2016

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The reinforcing steel industry is currently developing high-strength reinforcing bars with specified yield strengths of 80 and 100 ksi, due to increased demand for such grades in concrete construction. However, none of the higher steel grades are able to match the benchmark mechanical properties of grade 60 steel; with each high-strength variant diverging from benchmark behavior in different ways. There is concern that the less ductile higher grade reinforcing bars may fracture at the bends and may require larger bend diameters. Limited tests are available that investigate the relation between bend diameter and the ductility, or conversely the brittleness, of reinforcing bars at bends. No such tests exist for the newly developed high-strength reinforcement having yield strengths of 80 and 100 ksi. Bend/re-bend (or re-straightening) tests were conducted on grade 60 and higher grade reinforcing bars to investigate relations between bend diameters and bend performance. The tests were monitored using digital image correlation technology from which never-before recorded comparative measures were obtained. Test results indicated significant differences in bend performance between bars of varying grades, such that wider bend diameters may be necessary for certain higher grade bars.

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Chapter 1: Introduction

1.1 Motivation

There is an increasing need for higher strength reinforcing steel in seismic and non-seismic applications. A main driver for higher strengths is the need to reduce bar congestion in seismic designs and reduce material quantities generally. Steel manufacturers in the United States are currently developing reinforcing bars with yield strengths reaching 120 ksi and with varying mechanical and chemical properties. The new high-strength bars are being produced using varying methods, the most common of which are quenching and tempering, and micro-alloying. However, none of the higher steel grades in production are able to match the benchmark mechanical properties of grade 60 steel; with each high-strength variant diverging from benchmark behavior in different ways. There is concern that the less ductile higher steel grades may fracture at the bends and may require larger bend diameters. Anecdotal evidence has been reported of high-strength bars (HSRB) fracturing at the bends when dropped at a construction site (particularly in cold weather).

In this report, high-strength reinforcing bars (HSRB) are defined as steel reinforcing bars with yield strengths of 80 ksi or higher (i.e., grade 80 or higher).

1.2 Objectives and Scope

Efforts are underway to produce new ASTM specifications for HSRB to give steel mills a clear target to aim for in their production of HSRB. To complete the ASTM specifications for HSRB, bar bending requirements need to be revisited given recent evidence of HSRB fracturing at bends. Limited tests are available that investigate the relation between bend diameter and the ductility, or conversely the brittleness, of

reinforcing bars at bends. No such tests exist for the newly developed high-strength reinforcement having yield strengths of 80 and 100 ksi.

The main objectives of this study are to evaluate the performance under load of bends in HSRB satisfying ACI 318-14 (2014) and compare that performance with the benchmark performance of grade 60 bars bent in the same way and to same diameter.

To achieve project objectives, bend and re-bend tests, such as those specified in New-Zealand and United-Kingdom standards (BS 4449:2005+A2:2009), were conducted on HSRB and grade 60 bars. The tests provided a measure of the reserve strength and ductility of bar bends, which cannot be obtained using visual inspection of bend cracking, as described in ASTM A615 and A706 standards for reinforcing bars (BS 4449:2005+A2:2009). HSRB with varying manufacturing processes, grades, diameters, and bend diameters were tested. The range of parameters was selected to represent the most typical bar properties and manufacturing processes currently in production or development in the United States. All bars were fully strain-aged after bending and prior to re-bending, to represent typical conditions of bar bends in concrete structures. Strain aging tests were conducted to evaluate the effects of strain aging on all bar types considered, and to determine the required wait time after pre-straining or bending before re-bending could be performed.

CHAPTER 2: Background

2.1 Metallurgy

Three main production methods are currently used in the United States to produce HSRB. Each of these methods generates HSRB with differing mechanical and chemical properties. The three processes are quenching and tempering, micro-alloying, and manipulation of the microstructure using alloying and heat treatment. Steel bars produced through quenching and tempering typically exhibit relatively low tensile to yield strength (T/Y) ratios and relatively high strains at fracture. Steel bars produced by micro-alloying have a relatively high tensile to yield strength ratio and relatively high strains at fracture. HSRB produced using the third production method are the only ones with ASTM specifications (ASTM A1035 (2011)). These bars typically have large tensile to yield strength ratios but relatively low strains at fracture. The differences between the three production methods and the bar properties they produce are briefly discussed in this section.

2.1.1 Quenching and Tempering (QT)

The process of quenching and tempering (QT) consists of quenching the steel immediately after rolling and then allowing the bar to be tempered by the heat remaining in the core while gradually cooling. As a result, the QT process produces steel with mechanical properties that vary significantly between its inner core layer and its outer skin layer, with the inner core having a lower yield strength and more ductility than the outer layer. QT treated bars retain their yield plateau since they have not been strain hardened and, since the overall chemical composition has not been altered, they can be weldable if their chemistry satisfies ASTM A706 requirements. QT steel typically exhibits a low T/Y ratio on the order of 1.15 for grade 100 reinforcing bars. Slavin and Ghannoum (2015) provides more details about the QT process.

2.1.2 Micro-Alloying (MA)

Micro-alloying is a process that involves introducing small amounts of alloys in order to achieve the desired properties in steel bars. Vanadium is one of the alloys most commonly used to increase the strength of reinforcing bars. It increases strength and fracture toughness primarily due to inhibition of grain growth during heat-treatment and the precipitation of carbides and nitrides. The use of Vanadium can reduce the amount of carbon needed to achieve higher strengths and is therefore useful for achieving weldable HSRB. Micro-alloying can produce a marked yield point and a T/Y ratio larger than that from quenched and tempered steels (on the order of 1.25 for grade 100 reinforcing bars). Slavin and Ghannoum (2015) provides more details about the micro-alloying process.

2.1.3 Patented Microstructure Manipulation (MMFX)

The patented MMFX process involves manipulating the microstructure of steel to obtain the desired mechanical properties and strength. The process generates bars with stress-strain relations that do not have a well-defined yield point, exhibit a relatively high T/Y ratio, but have relatively low fracture elongations. The MMFX steel bars satisfy the ASTM A1035 specifications. The A1035 specifications maintain the same bend diameter for testing bar bends as in the ASTM A615 and A706 specifications used for the vast majority of bars currently in production in the United-States. ACI 318-14 (2014) allows the use of A1035 grade 100 bars in confinement applications and requires them to be bent at the same diameter as other steel grades including grade 60.

2.2 Strain Aging

Strain aging is defined as the process by which steel strained beyond its elastic limit undergoes time dependent changes in its mechanical properties. Typically, reinforcing bars strained beyond their elastic limit will, over time, see an increase in their tensile

strength and a decrease in their ductility as illustrated in Figure 1. Strain aging is also proven to affect the brittle transition temperature in steel (G.T. Van Rooyen, 1986). Factors affecting strain aging include the steel composition, temperature, and the time elapsed since large strains were incurred. Strain aging is mostly attributed to nitrogen reallocation within the steel matrix (G.T. Van Rooyen, 1986). Higher temperatures accelerate this process; hence strain aging occurs much faster in warmer regions. Typically, most of the effects of strain aging in steel reinforcing bars will occur within a few months after inelastic strains are incurred (G.T. Van Rooyen, 1986).

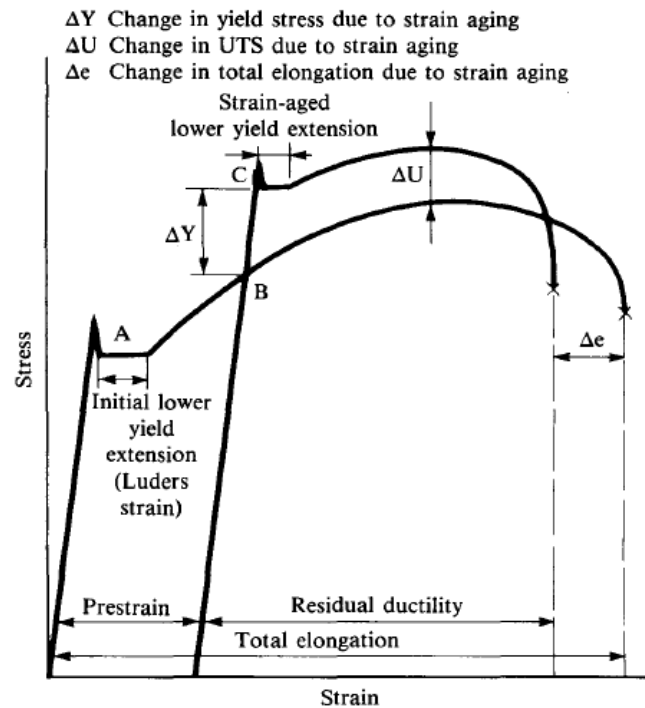


Figure 1: Typical stress-strain curves showing the effects of strain aging (G.T Van Rooyen, 1986)

As reinforcing bars are bent, they experience large inelastic strains. Bar bends are therefore prone to strain aging embrittlement, which may cause them to fracture prematurely and limit their ability to sustain inelastic deformations during structural loading.

Research done by G.T Van Rooyen (1986) and Rashid (1976) suggests that micro-alloyed steel including titanium and vanadium can lower the effects of strain aging on steel bars. Such alloys have properties that allow them to bond with the nitrogen in the composition to form nitrides. These reactions limit the amount of free nitrogen throughout the steel that is attributed to strain aging effects.

2.3 Bend Tests

Three main categories of experimental tests are useful for investigating the behavior of bends in reinforcing bars, with each category of tests geared to answer a particular set of questions:

1 - Visual inspections of bends (ASTM bend tests)

2 - Bend/re-bend tests

3 - Bend tests in concrete

1 - ASTM reinforcing bar specifications (such as A615, A706, and A1035) specify the following bending requirement “The bend test specimen shall withstand being bent around a pin without cracking on the outside of the bend portion.” The required bend test therefore involves bending bars to 180° (or 90° for #14 and larger bars) at a specified pin bend diameter. A visual inspection is then performed to identify cracking at the bend. If no cracking is visually observed, a specimen is deemed to pass the bend test. The pin diameters specified in ASTM bend tests are tighter than those used in construction, as specified by ACI 318-14 or the CRSI handbook, and therefore provide some degree of safety against observing cracking in bends during bar fabrication. However, while this test is simple to perform, it does not provide a measure of the reserve strength and ductility of bar bends, as a load-test can. It is possible that micro-cracking not visible to the eye may compromise the performance of bars in-situ.

2 - Bend and re-bend tests have been performed in New-Zealand (AS/NZS 4761:2001; Hopkins and Poole (2008) and the United-Kingdom (BS4449 (2005))). In these tests, bar coupons are bent to the required angle and bend diameter (Figure 2), and then straightened at either quasi-static or dynamic loading rates. For grade 60 bars, work hardening increases steel strength at the bends and typically causes the coupons to fracture away from the bends in a ductile manner. However, if bars have limited ductility such as HSRB, strain demands at the bends may cause cracks, which can make bends weaker than the unbent portions of the bars and more susceptible to brittle fracture. If a bar fails in a brittle manner at a bend, it is considered to have failed the bend/re-bend test. If, however, a bar fails in a ductile manner, then it is deemed to have passed the test. This type of test has the advantage of putting bar-bends under load and therefore provides a direct measure of the strength and ductility performance of bar-bends.

Hopkins and Poole (2008) conducted bend and re-bend tests on newly introduced grade 500E (~72 ksi) bars in New Zealand and Australia. The study accounted for strain aging and explored the effects of cold temperature on the performance of bends. The study tested bars produced using micro-alloying (MA), as well as quenching and tempering (QT). Test results confirmed that current bar bend diameters used in New Zealand were adequate for that grade of steel, regardless of the manufacturing process. A marked worsening of the performance of bar-bends was observed at temperatures below -10° Celsius.

It should be noted that the bend/re-bend tests apply larger demands on the bar bends than they would normally see in a concrete structure. For this reason, it is best to compare the bend/re-bend performance of HSRB to that of grade 60 bars, which have been used for decades and have shown adequate performance in concrete members.

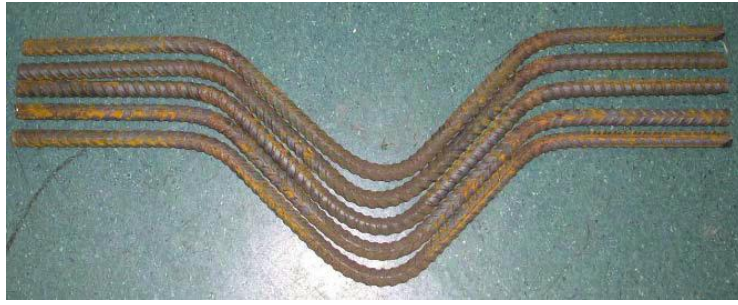


Figure 2: Picture of bend/re-bend test coupons (NIST GCR 13-917-30)

3 - Bends in reinforcing bars can also be tested in concrete. In such tests, the interaction between the concrete and bar-bends can be investigated. Simplified versions of the test include embedding a hooked bar into a concrete block and pulling on it until failure. Possible failure modes that can be expected in block tests include: bar fracture outside the block where demands on the bar are highest, bar failure inside the block closer to or at the bend, or splitting of the concrete block. Such tests, however, may not expose bends to the worst loading they could experience in a structure, as the surrounding concrete can relieve the bends of some load. In contrast, some of the worst loading on bar-bends can arise in confinement applications, where an expanding concrete core partially straightens hoop bends while applying high tensile loads to them. Another critical application for bar-bends is in damaged regions, where bond to concrete and its beneficial effects on bends are reduced (e.g., joints under severe seismic loading, or severely cracked regions).

Nevertheless, tests of bar bends in concrete members are essential for validating the adequate performance of bar bends in HSRB. However, such tests are expensive to conduct and do not easily lend themselves to the task of determining minimum bend diameters while exploring the numerous variables that affect the performance of bar-bends.

CHAPTER 3: Experimental Program

3.1 Overview of Program

The experimental program consisted of three types of tests.

1. Bend/re-bend Tests
2. Strain Aging Tests
3. Monotonic Tests

3.1.1 Bend/re-bend Tests

Objectives: Bend/re-bend tests were conducted to quantify residual strength and elongation capacities under load in bar bends and compare high-strength reinforcing bar bend performance with that of grade 60 bar bends.

3.1.1.2 Specimen Details and Preparation

The bend/re-bend specimens were similar to those used in New-Zealand by Hopkins and Poole (2008) and described in BS 4449:2005+A2:2009 (2005). Bar specimens were constructed by bending straight coupons into a “V” shape having two 45 degree bends and one 90 degree bend (Figure 2). Coupons were bent by a local fabricator according to typical bending practices using an RMS Arnold Bender (Figure 3). Bars were bent about their weak axis, with the longitudinal ribs facing vertically in the bender. Bending was conducted at a room temperature of about 20°C. Bars were then left to strain age prior to re-bending them in tension until fracture in a uniaxial testing machine. Tolerance templates were used to ensure that specimens were bent accurately (Figure 4). The internal bend diameters for the first batch of bar specimens were the smallest specified in ACI 318-14. The dimensions of bend/re-bend coupons with those bend diameters are presented in Figure 5. In subsequent bending, some #5 bar bend diameters were increased to $5d_b$ and $6d_b$ (with d_b = bar nominal diameter).



Figure 3: RMS Arnold Bender used to bend specimens



Figure 4: Verifying bend specimen tolerance

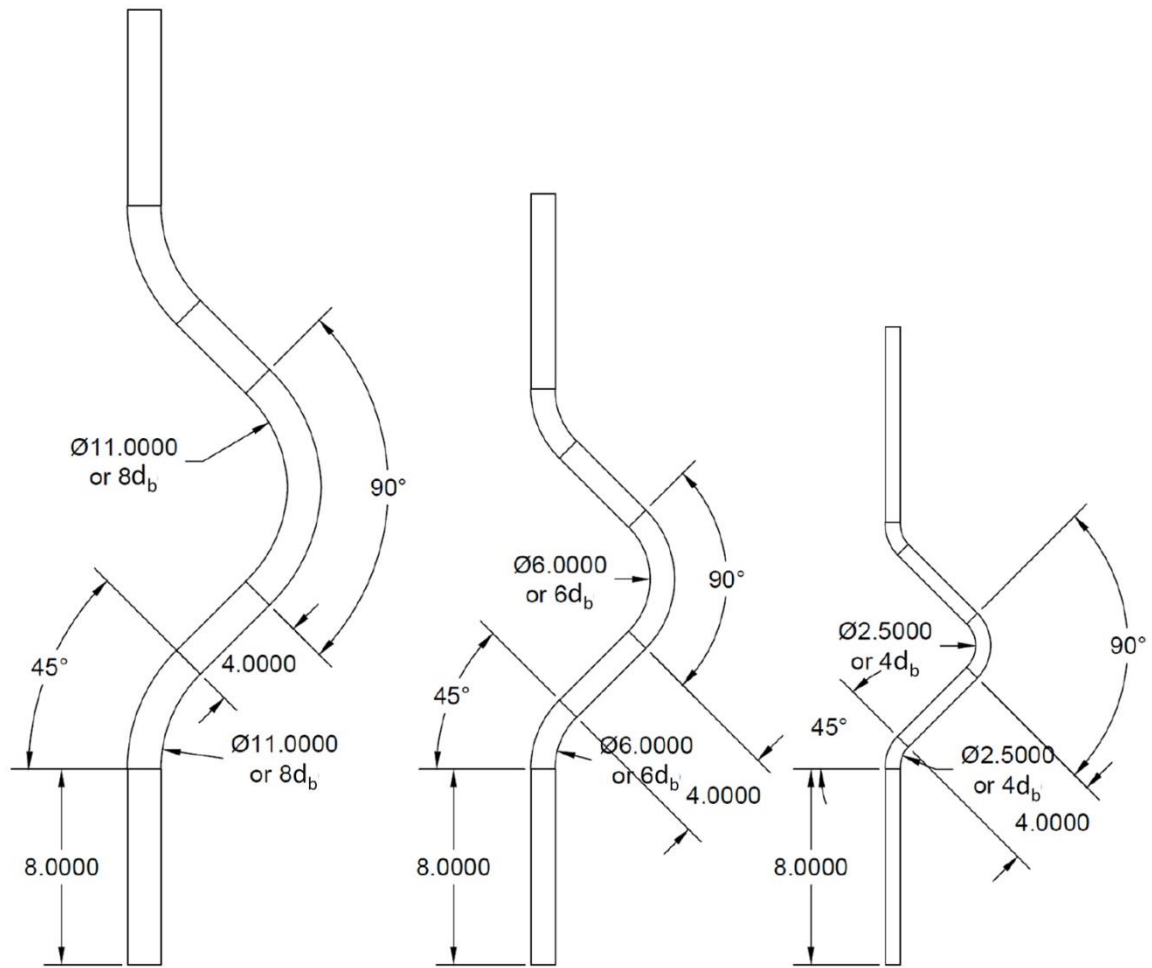


Figure 5: Specimen dimensions for #11, #8 and #5 bars with ACI 318-14 minimum bend diameter (d_b = bar nominal diameter)

3.1.1.3 Controlled Test Parameters

- a- **Steel Grade and Specifications:** A706 (high ductility) and A615 (lower ductility) grade 60 bars were tested to provide benchmark performance. Grade 60 bars were obtained from three mills utilizing the two main production techniques for that grade: micro-alloying (MA) and quenching and tempering (QT). HSRB produced using the main three production methods in the U.S. were also tested:

MA (Micro-Alloying), QT (Quenching and Tempering), and MMFX (ASTM A1035). The main focus in HSRB was on grade 100 steel (having a yield strength equal to or higher than 100 ksi) but limited tests were conducted on grade 80 bars as well. In all, steel bars from four manufacturers were tested:

- a. Manufacturer 1 (M1): Micro Alloyed Steel (MA)
 - b. Manufacturer 2 (M2): Patented Microstructure MMFX (ASTM A1035)
 - c. Manufacturer 3 (M3): Combination of Quench and Tempering and Micro Alloying (QT)
 - d. Manufacturer 4 (M4): Combination of Quench and Tempering and Micro Alloying (QT)
- b- **Bar Size:** Three bar sizes were tested in this study covering a common range of sizes used in construction. The bars sizes used were #5, #8, and #11.
- c- **Bend Diameters:** Bar specimens were first bent to the current ACI 318-14 minimum bend diameters. For #11 bars, the internal bend diameter was about $8d_b$. For #8 bars, the internal bend diameter was about $6d_b$. For #5 bars, the minimum bend diameter for transverse reinforcement was initially selected, which is $4d_b$. However, after observing poorer performance in #5 HSRB at that bend diameter, #5 bars were bent and tested with $5d_b$ and $6d_b$ bend diameters. The latter bend diameter of $6d_b$ corresponds to the ACI 318-14 minimum bend diameters for #5 longitudinal or other bars in tension. It is noteworthy that the selected bar sizes (#5, #8, and #11) correspond to the largest size with a given bend diameter before the next bigger size requires a larger bend diameter in ACI 318-14.

Due to bending pin availability and spring back in the bending process, final internal bend diameters were close to but not exactly equal to their target

values. Table 1 lists the target, as well as the archived bend inside diameters for the 90 degree bends in the specimens. The table also lists the pin diameters used during bending.

Table 1: Measured d_b for various bar sizes vs d_b from pin used for bending

	Target d_b	Mean d_b Measured	Pin Diameter (in)	d_b Pin	% Difference
#11 ($8d_b$) - 4 Samples	8	7.82	10.0	7.30	7.1%
#8 ($6d_b$) - 4 Samples	6	5.64	5.0	5.00	12.7%
#5 ($6d_b$) - 2 Samples	6	5.39	3.0	4.80	12.3%
#5 ($5d_b$) - 3 Samples	5	4.80	2.5	4.00	20.0%
#5 ($4d_b$) - 3 Samples	4	3.59	2.0	3.20	12.3%

A mean d_b of varying sample sizes was measured for all bend diameters (Table 1). Overall the mean measured d_b was lower than the target d_b and higher than the d_b of the pin by ranging from 7-20%. For all the bars there was a noticeable % difference from d_b pin to d_b measured which was caused by spring back of the bars during bending. Bending pins were only offered in $\frac{1}{2}$ " increments, hence the pin diameters that were chosen such that the d_b measured was as close to target d_b as possible but not larger.

3.1.1.4 Fixed Parameters

- a- **Bend Angle:** A primary bend angle of 90° was tested in this study. Other bend angles were not included in the scope of this project.
- b- **Loading Rate** was applied quasi-statically to re-bend tests in this project.
- c- **# Of Specimens:** At least 3 bar specimens per type were tested.

- d- **Strain Aging:** Bar types sensitive to strain aging (as determined in the strain aging tests) were bent and allowed to strain age prior to being subjected to the re-bend tests. This was done to account for the possible deleterious effects of aging on bar ductility, and to reproduce actual in-situ conditions of bends in concrete structures.
- e- **Temperature:** The focus of this project was on assessing the bend performance of HSRB at relatively warm ambient temperatures (20 to 25 °C). This evaluation is a necessary first step prior to bending and testing bars in cold climate at or below their brittle transition temperatures. Cold temperature bending and testing should be evaluated in future work.
- f- **Bending Equipment:** An RMS Arnold Bender with the capability to bend at 16 RPM and 14RPM was used for all specimens.
- g- **Bending Rate:** All specimens were bent at 16 RPM.

3.1.1.5 Instrumentation

The load-cell in the universal testing machine provided readings of the applied load. Strains and deformations of the bars were obtained during testing using a high-resolution optical measurement system reported by Sokoli et al (2014). Targets were applied along the four straight portions of bent bar specimens (Figure 6). The locations of the targets were tracked using the optical system while the bars were being tested (Figure 6).

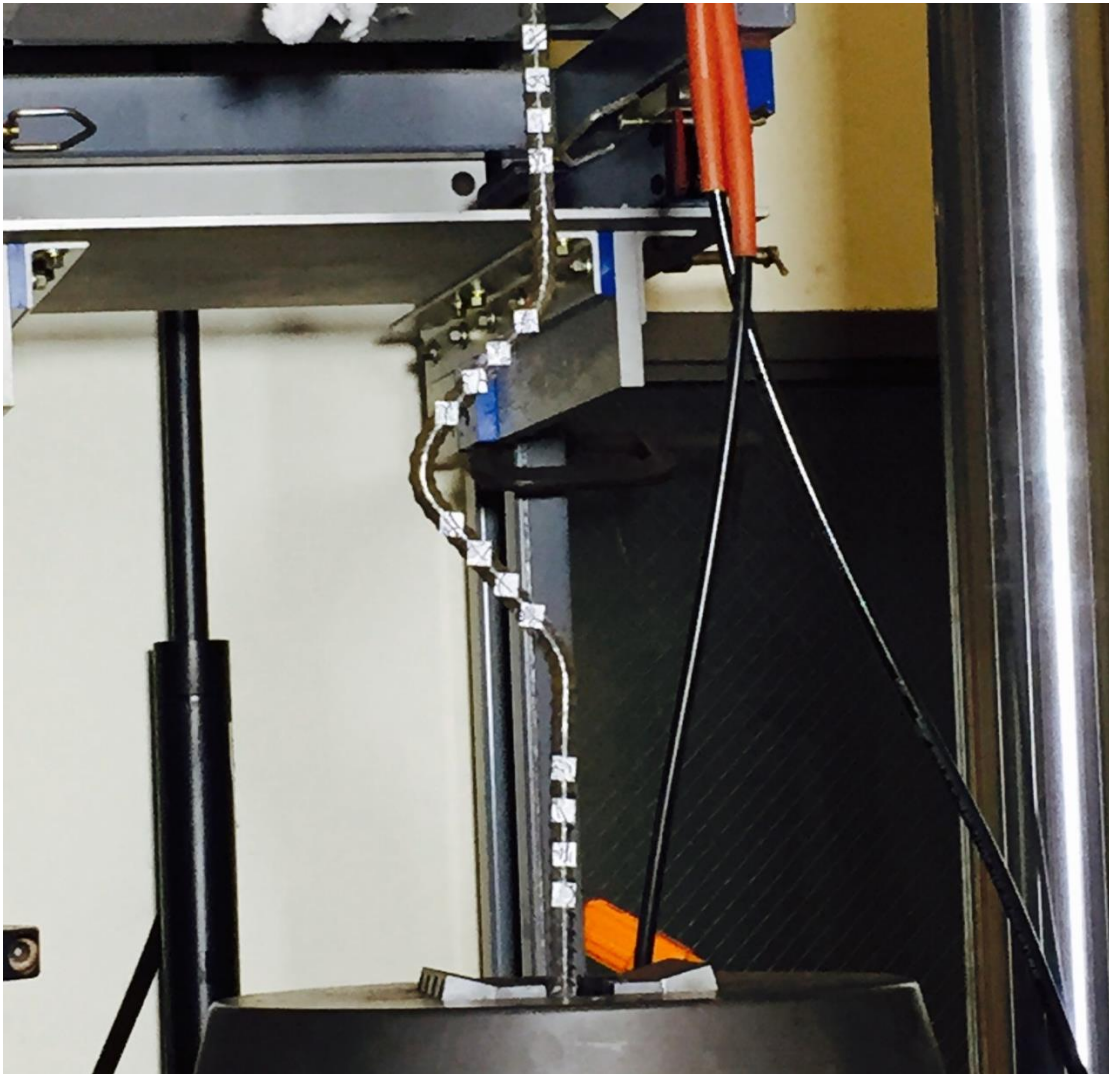


Figure 6: Example of bend/re-bend specimen under testing with targets

3.1.1.6 Test Protocol

Bar were gripped at each end using hydraulic grips that were 6 inches long. The loading during re-bend tests consisted of three loading rates. The loading protocol started with a low loading rate to pick up initial load in the specimens. This was done in order to obtain sufficient data in cases of bar fracture at low forces. Once specimens carried a larger force (~ 10% of yield) the loading rate was increased until the bars were almost straight. Then the rate was again lowered to observe fracture in more detail. All three bar sizes were loaded with the same rates presented in Table 2.

Table 2: Re-bend test loading rates

Loading Rates (in./s)	
Initial Loading	0.0050
Straightening	0.0150
Fracture	0.0075

3.1.2 Strain Aging Tests

Objectives: To quantify strain aging effects on all the bars tested in the bend/re-bend tests and identify the duration after which most strain aging effects level off.

Strain-aging tests were conducted on straight #5 bar coupons for all steel types. All bars, except those satisfying ASTM A1035, were strained in tension to a predetermined strain value of 0.04 in a universal test machine and then unloaded. A1035 bars were strained in tension only to a strain of 0.02 due to their relatively low uniform elongation values. Uniform elongation is defined as the bar elongation at peak stress. Tension tests were then performed on the pre-strained bars immediately after pre-straining, one month after pre-straining, and three months after pre-straining. Strain aging was allowed to occur at room temperature ($\sim 20^{\circ}\text{C}$). The bars were strained in the strain-aging tests at the strain rate used in the monotonic tension tests.

3.1.3 Monotonic Tests

Monotonic tension tests were performed on straight specimens to identify the material properties of the steel bars. The tests followed the procedures specified in ASTM A370 (ASTM Standard A370-15), with strains measured over a gauge length of 8 inches.

3.2 Specimen Nomenclature

For each test type, the following nomenclature is used to identify specimens:

Manufacturer#_Grade_Bar Type or Specification_Diameter (in eighth of an inch)_Test unique identifier

e.g.: (M1_Gr60_A706_5_01 or M1_Gr100_MA_5_01)

For strain aging tests the following nomenclature was used:

Manufacturer#_Grade_Bar Type or Specification_Diameter (in eighth of an inch)_MonthsAged_Test unique identifier

e.g.: (M1_Gr60_A706_5_3Mo_01)

CHAPTER 4: Test Results and General Observations

4.1 Monotonic Tests

Monotonic tests were conducted on three or more coupons for each bar type used in the bend/re-bend test matrix. The mechanical properties and typical stress-strain relations of each bar type are summarized in Table 3 and Figure 7 to Figure 11. Uniform elongation measures were obtained by following the ASTM E8 procedures (ASTM Standard E8/E8M-15a).

Table 3: Summary of mean material properties calculated from monotonic tension tests

Bar Size	Manf.	Grade	Yield Strength f_y (ksi)	Tensile Strength f_u (ksi)	T/Y Ratio	Uniform Elongation ϵ_{un} (%)	Fracture Elongation ϵ_f (%)
#11	1	A706 60	64.3	93.2	1.45	12.5%	21.9%
		A706 80	81.7	111.2	1.36	10.3%	18.2%
		MA 100	110.4	139.6	1.26	8.8%	12.7%
		A615 60	63.2	104.0	1.64	11.2%	17.9%
		A615 80	80.5	121.1	1.50	9.1%	14.4%
	2	A1035 100	125.0	162.1	1.30	4.9%	11.7%
	3	A706 60	77.4	102.8	1.33	9.1%	14.5%
		A706 80	83.1	109.7	1.32	9.1%	13.8%
	4	A615 60	63.6	90.7	1.43	12.1%	17.1%
		A706 60	67.5	95.8	1.42	11.5%	16.0%
#8	1	A706 60	60.5	90.0	1.49	11.5%	18.9%
		A706 80	84.4	114.1	1.35	9.8%	16.4%
		MA 100	99.0	125.2	1.27	8.9%	13.0%
		A615 60	63.7	101.3	1.59	10.7%	16.2%
		A615 80	84.4	123.5	1.46	9.2%	14.6%
	2	A1035 100	131.4	164.3	1.25	5.2%	10.8%
	3	A706 60	80.9	101.7	1.26	9.0%	14.9%
		A706 80	81.6	104.0	1.27	8.9%	14.6%
		QT 100	98.7	126.0	1.28	7.2%	9.9%
	4	A615 60	68.1	95.8	1.41	12.0%	18.3%
		A706 60	66.7	90.9	1.36	12.3%	18.5%
#5	1	A706 60	65.7	93.9	1.43	10.5%	14.7%
		A706 80	86.5	115.2	1.33	9.5%	13.8%
		MA 100	113.0	135.1	1.20	8.2%	11.9%
		A615 60	63.0	97.9	1.55	11.2%	16.5%
		A615 80	81.8	112.9	1.38	9.9%	13.9%
	2	A1035 100	125.6	163.6	1.30	5.4%	9.5%
	3	A706 60	81.6	99.7	1.22	8.8%	12.7%
		A706 80	83.3	102.7	1.23	8.8%	12.6%
		QT 100	90.0	129.7	1.44	6.9%	8.4%
	4	A615 60	80.2	102.9	1.28	10.3%	15.0%
		A706 60	66.0	90.8	1.38	12.1%	18.4%
		QT 100	106.1	125.6	1.18	7.6%	10.7%

A706 Grade 60

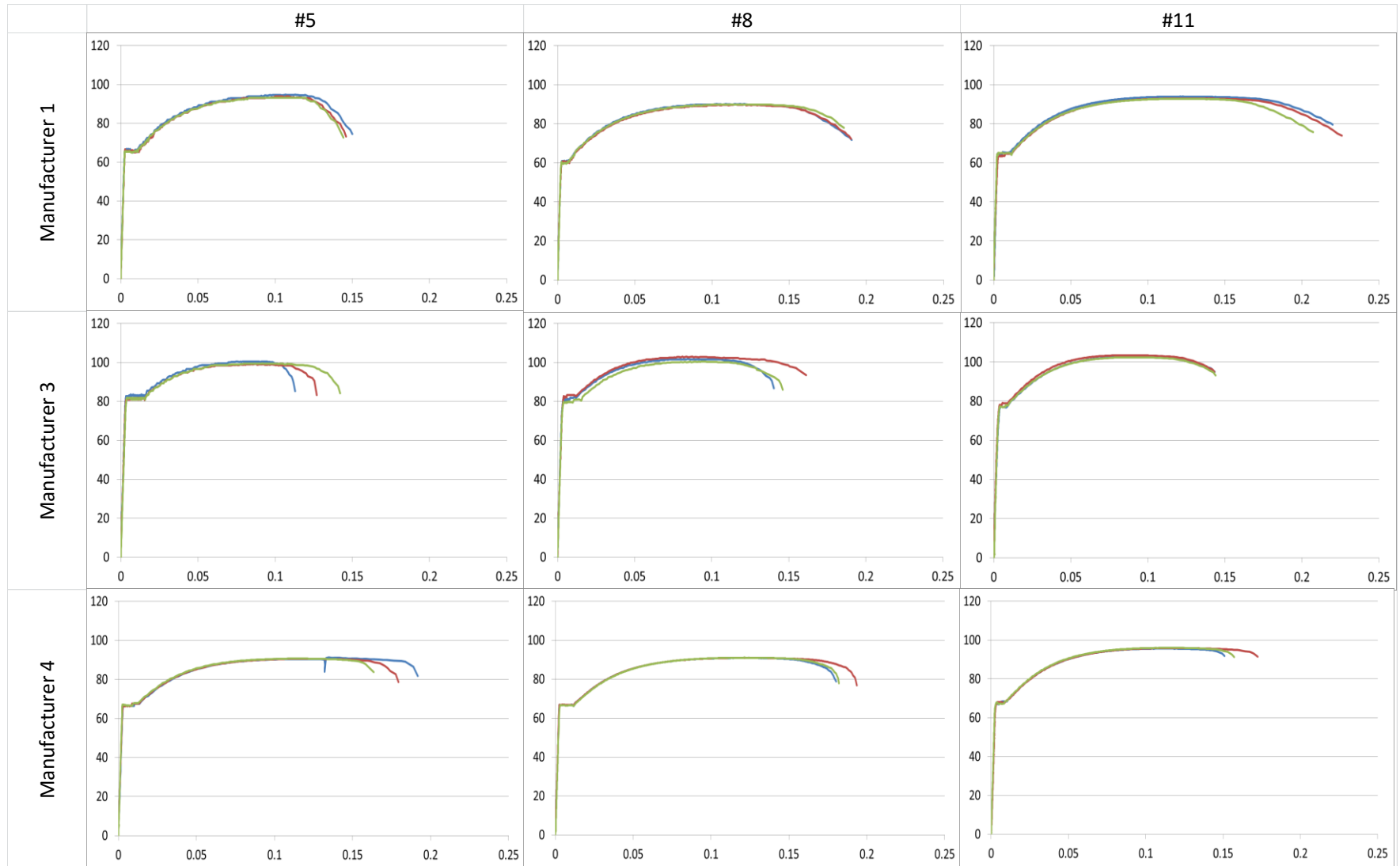


Figure 7: Stress-strain curves from monotonic tests of grade 60 A706 bars

A615 Grade 60

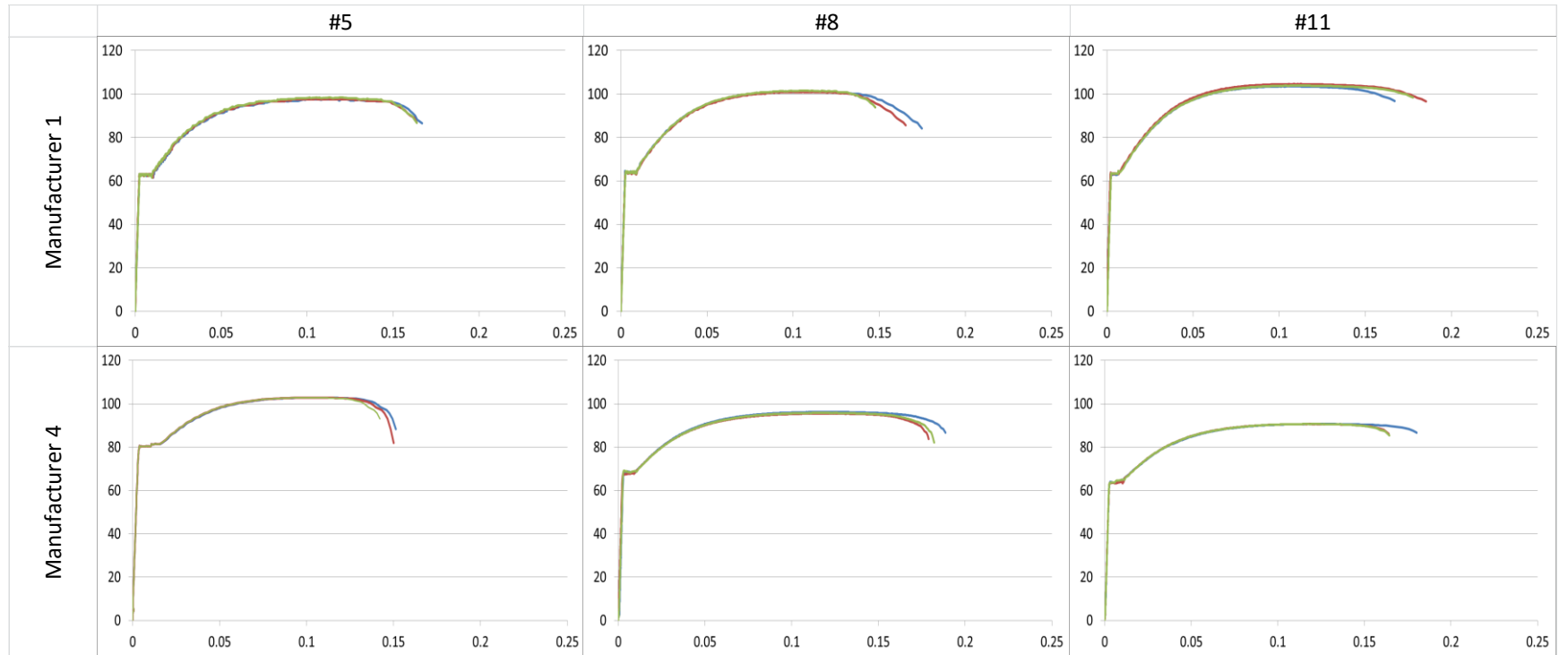


Figure 8: Stress-strain curves from monotonic tests of grade 60 A615 bars

A615 Grade 80

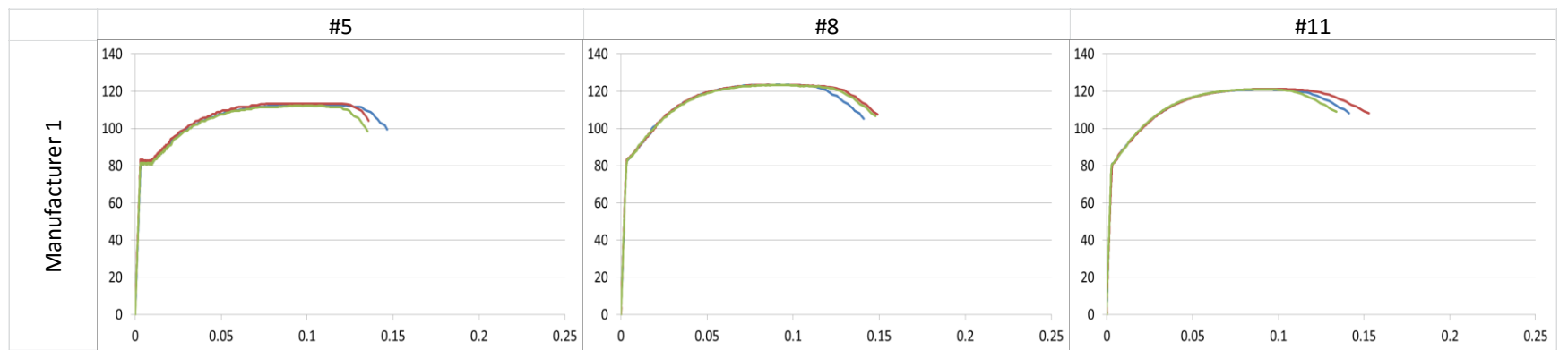


Figure 9: Stress-strain curves from monotonic tests of grade 80 A615 bars

A706 Grade 80

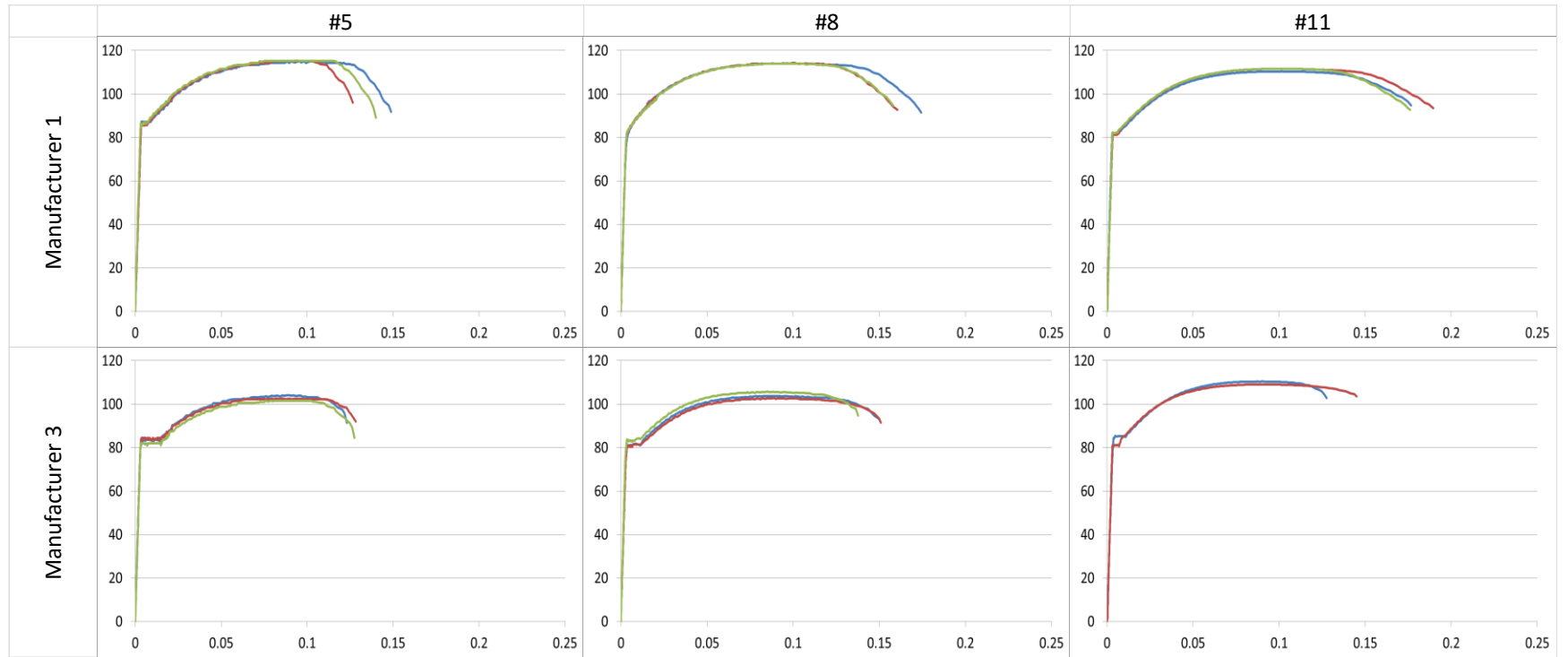


Figure 10: Stress-strain curves from monotonic tests of grade 80 A706 bars

Grade 100

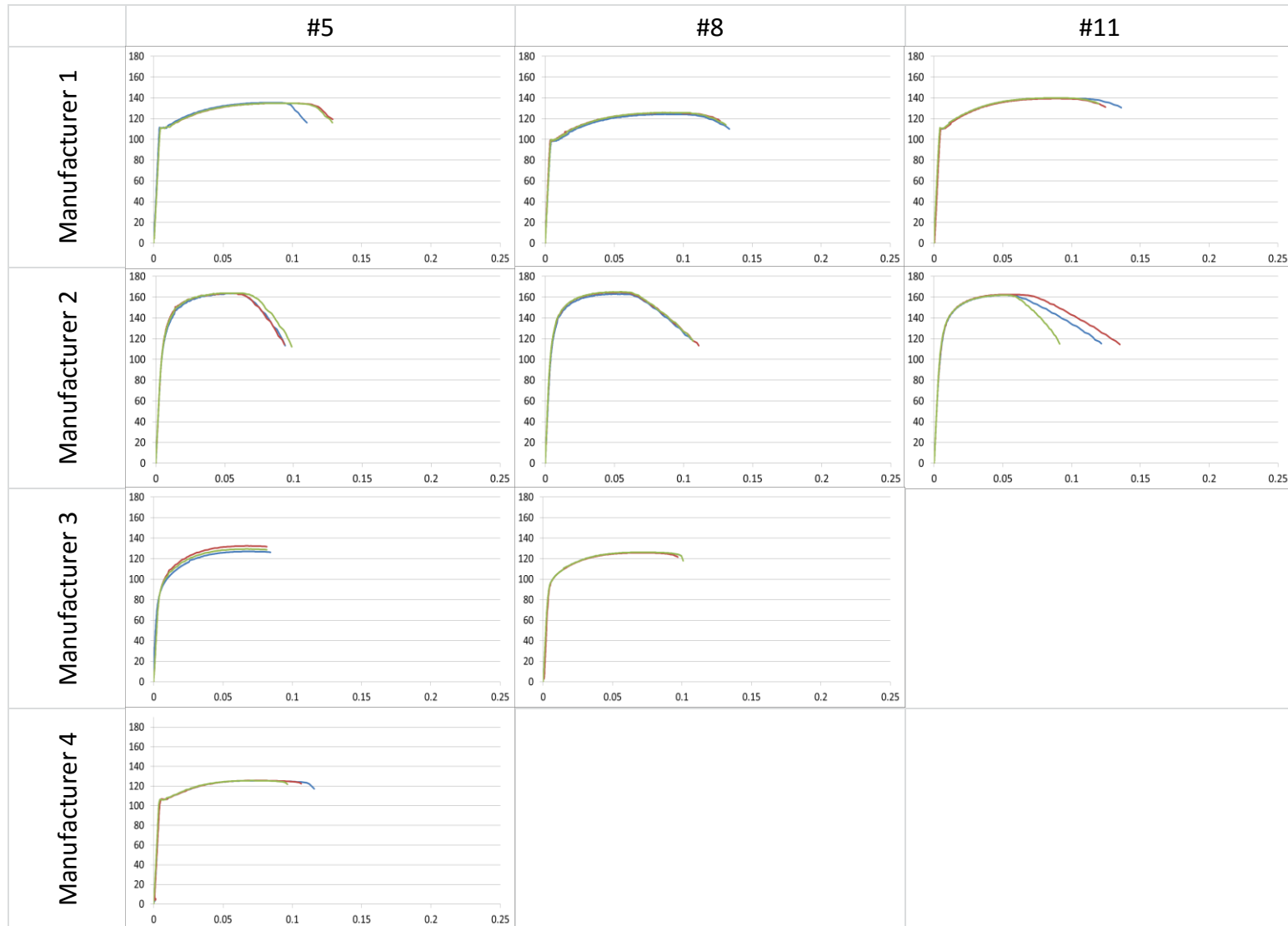


Figure 11: Stress-strain curves from monotonic tests of grade 100 bars

4.1.1 Summary of Observations

The yield strengths of grade 60 A706 bars tested in this study ranged from 63-77 ksi. The fracture strains of those bars were in the range of 15% to 22%. The yield strengths of grade 60 A615 bars tested in this study were around 63 to 64 ksi with the exception of Manufacturer 4's #5 bars that came in at a yield strength of 80.2 ksi. The fracture strains of grade 60 A615 bars were in the range of 15% to 19%.

Grade 80 A706 bars had yield strengths ranging from 81 to 87 ksi. The fracture strains of those bars were in the range of 13% to 19%. Grade 80 A615 bars were only obtained from Manufacturer 1 and had yield strengths ranging from 80 to 84 ksi for all sizes and fracture elongations ranging from 14% to 15%.

Grade 100 bar from Manufacturers 1 and 4 exhibited a yield plateau and had yield strengths in the range of 105 to 110 ksi, with fracture strains ranging from 11 to 13%. Bars from Manufacturers 2 and 3 showed no yield plateau. Yield strengths for those bars were obtained using a 0.2% strain offset method (ASTM A1035, 318-14). Yield strength values for Manufacturer 2 ranged from 125 to 131 ksi while those for Manufacturer 3 ranged from 90 to 99 ksi. Manufacturer 2 grade 100 bars exhibited fracture strains from 10 to 12% whereas Manufacturer 3 grade 100 bars exhibited fracture strains from 9% to 10 %. Manufacturer 2 grade 100 bars differ from other bars as they exhibited a relatively shallow descending relation between stress and strains past their uniform elongation.

4.2 Strain Aging Tests

4.2.1 Strain Aging Test Data

Strain aging tests were conducted on #5 bars of all types used in bend/re-bend tests. Figure 12 to Figure 14 present typical stress-strain relations obtained from the strain aging tests. The arrows in the figures are used to show the difference from non-strain aged to strain aged results. As expected based on past research (Rashid, 1976), grade 80 and 100 bars produced using micro-alloying with Vanadium (Manufacturer 1) exhibited limited strain aging compared with grade 60 bars from the same manufacturer (Figure 12). Bars from Manufacturer 4, which contained relatively small amounts of micro-alloys, exhibited more pronounced strain aging effects as indicated by apparent gains in tensile strength and decreased ductility (Figure 13). Figure 14 highlights how different manufacturing processes result in different strain aging results.

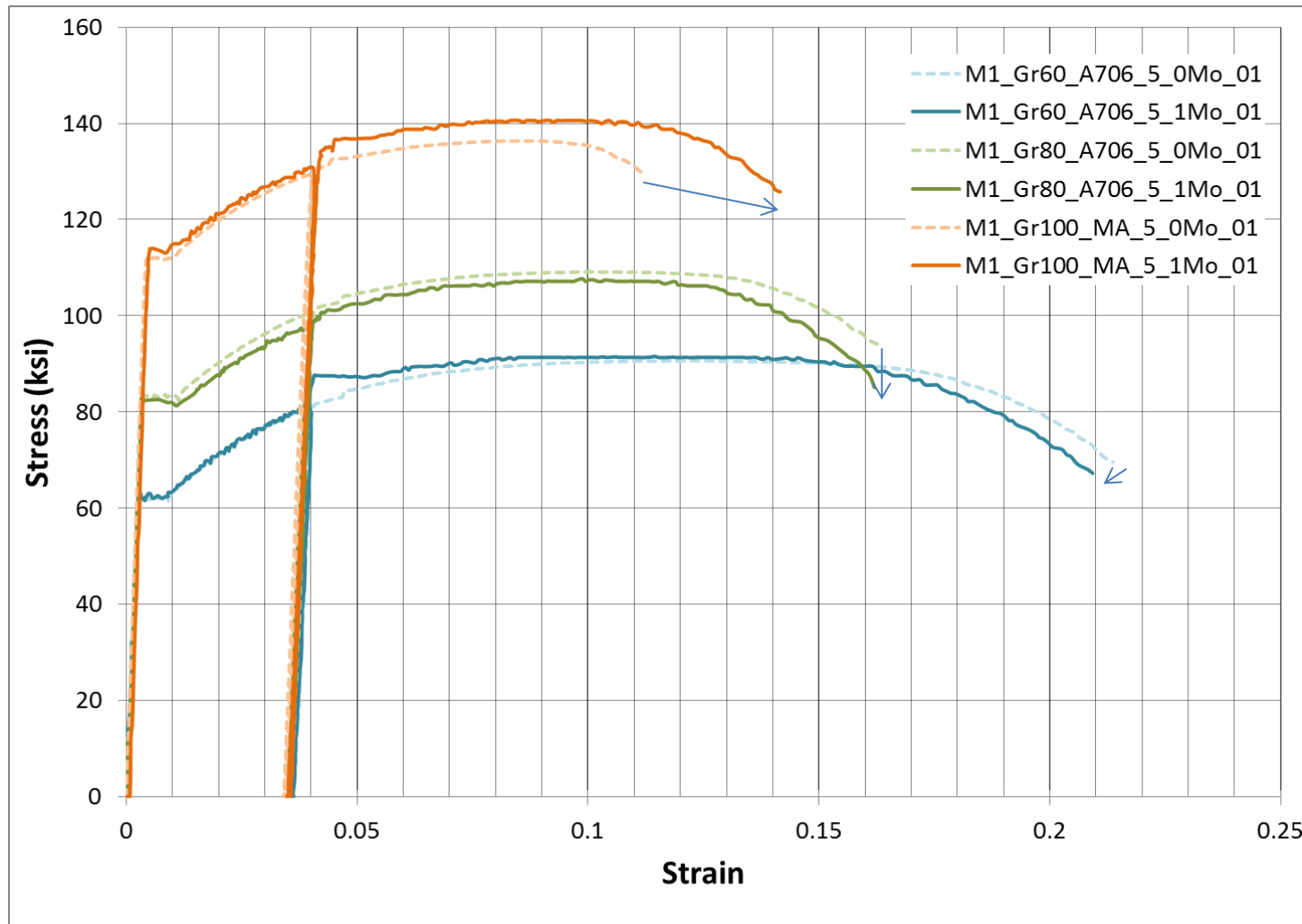


Figure 12: Stress-strain curves for grades 60 and 80 A706 and grade 100 MA bars from Manufacturer 1, not aged and strain aged 1 month

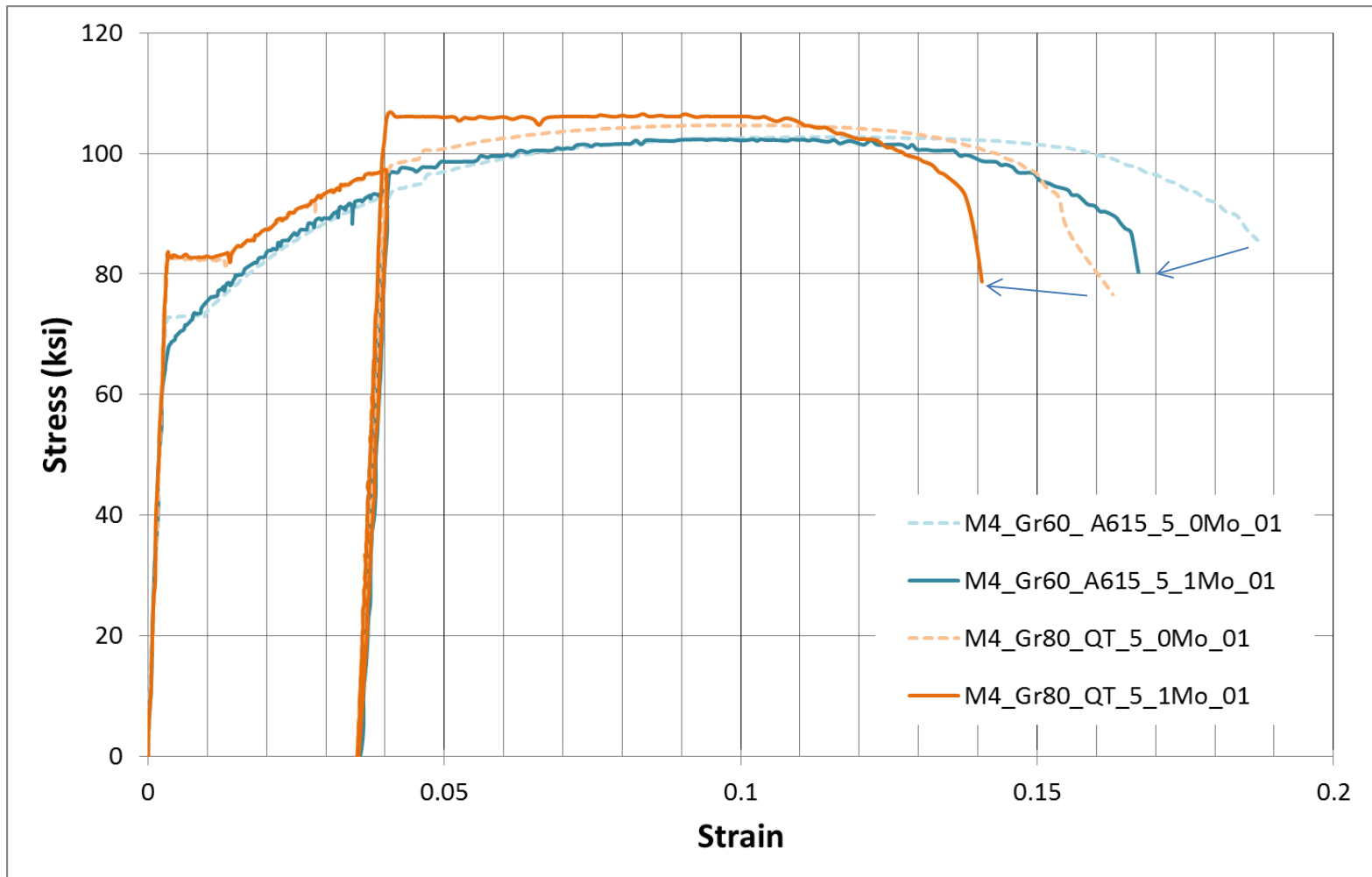


Figure 13: Stress-strain curves for grade 60 and 80 A706 bars from Manufacturer 4, not aged and strain aged 1 month

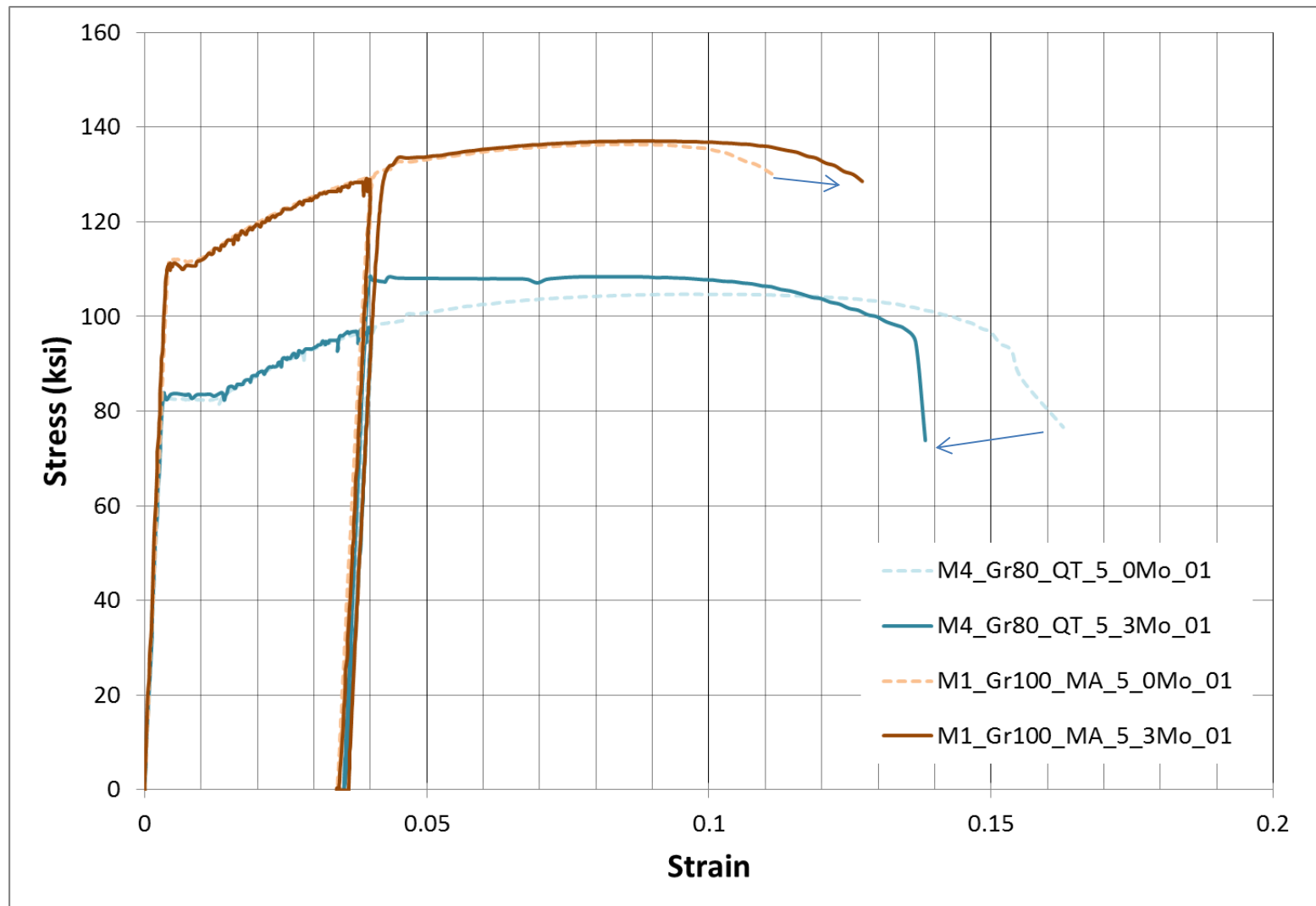


Figure 14: Stress-strain curve comparisons between grade 80 QT from Manufacturer 4 and grade 100 MA from Manufacturer 1, not aged and strain aged 3 months

4.2.2 Strain Aging Performance Measures

Two parameters were used to quantify the effects of strain aging, $(\Delta\sigma/f_y)$ and $(\epsilon_{\text{fractureA}}/\epsilon_{\text{fracture}})$. $\Delta\sigma$ is the difference between the apparent yield point upon reloading after specimen aging and the stress at unloading as seen in the Figure 15. $\Delta\sigma$ was normalized with respect to the measured yield strength of the bars, f_y , to for comparison between different grades. The strain at bar fracture of aged bars normalized by the strain at fracture prior to aging $(\epsilon_{\text{fractureA}}/\epsilon_{\text{fracture}})$ was also used to assess the severity of strain aging. The strains were measured over an eight inch gauge length.

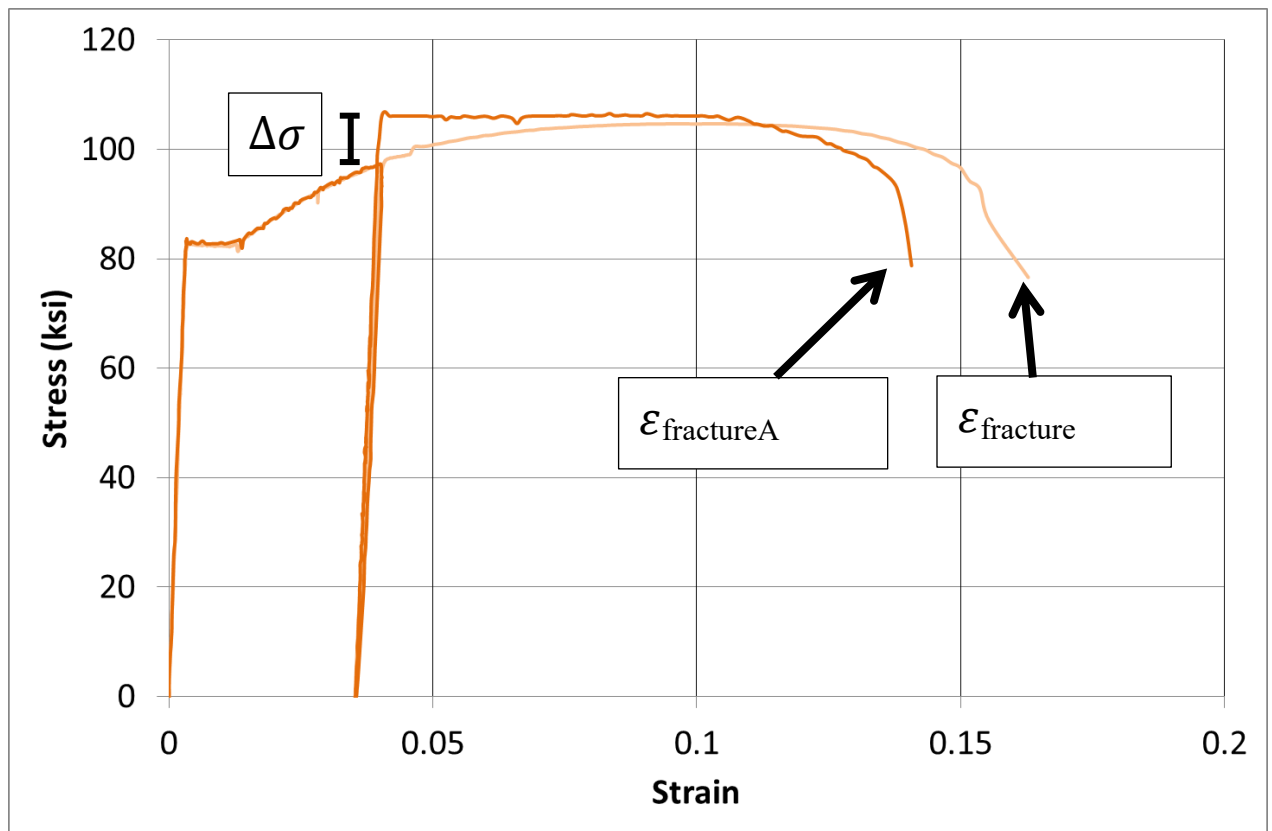


Figure 15: Apparent yield point and loss of elongation after strain aging

4.2.3 Strain Aging Results

Table 4 summarizes the strain aging results averaged over the coupons tested per bar type. At least three coupons were tested per bar type. The concentration of Vanadium was found to be highly correlated with strain aging effects, namely the increase in apparent yield strength and the reduction in fracture elongation. Table 4 also summarizes the Vanadium concentrations for each bar type.

Table 4: Summary of strain aging test results

Bar Type	Measured Yield Stress f_y (ksi)	Percent concentration of Vanadium (V %)	$\Delta\sigma/f_y$ after 1Mo of Strain Aging	$\Delta\sigma/f_y$ after 3Mo of Strain Aging	$(\epsilon_{fractureA} / \epsilon_{fracture})$ after 1Mo of Strain Aging	$(\epsilon_{fractureA} / \epsilon_{fracture})$ after 3Mo of Strain Aging
M4_Gr60_A706_5	73.2	0.038	0.05	0.08	0.89	0.90
M4_Gr80_A615_5	82.8	0.002	0.11	0.13	0.85	0.85
M1_Gr60_A706_5	63	0.024	0.10	0.10	0.87	0.82
M1_Gr80_A706_5	82.4	0.098	0.03	0.03	0.95	0.91
M1_Gr100_MA_5	112	0.355	0.02	0.04	1.17	1.08
M1_Gr60_A615_5	63.34	0.012	0.13	0.13	0.96	0.99
M1_Gr80_A615_5	83.5	0.063	0.02	0.02	0.98	1.07

4.2.3.1 Effects of Strain Aging Duration

Figure 16 and Figure 17 illustrate the variation of $(\Delta\sigma/f_y)$ and $(\epsilon_{fractureA} / \epsilon_{fracture})$ with strain aging duration. Dashed lines represent grade 60 bars, shaded lines represent grade 80 bars and solid lines represent grade 100 bars. As can be seen in the figures, the majority of strain aging effects occur within the first month with limited changes observed thereafter for all bar types. Based on these findings, bend/re-bend tests were conducted on bars starting one month after the initial bending was conducted.

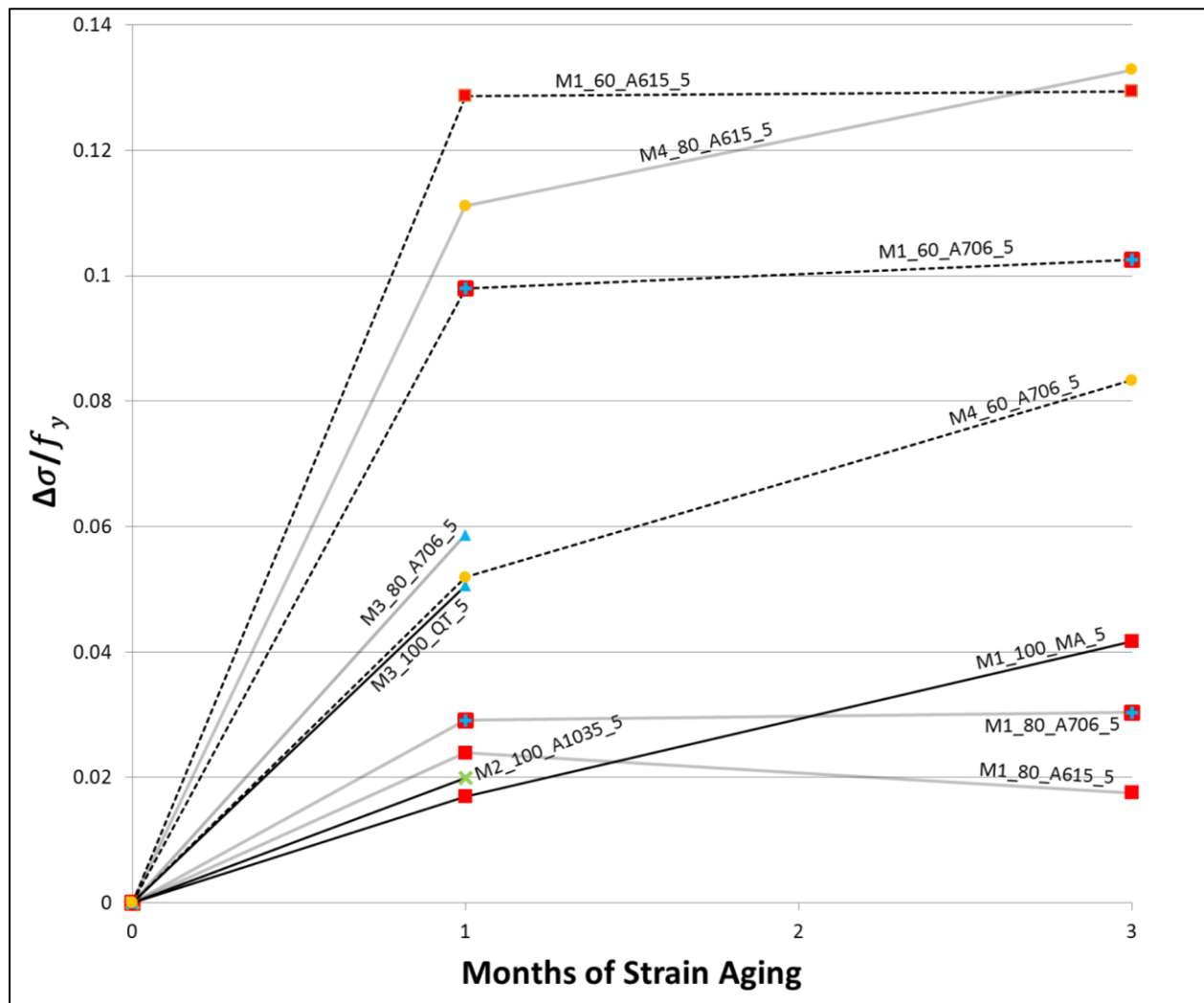


Figure 16: Normalized $\Delta\sigma$ vs strain aging duration

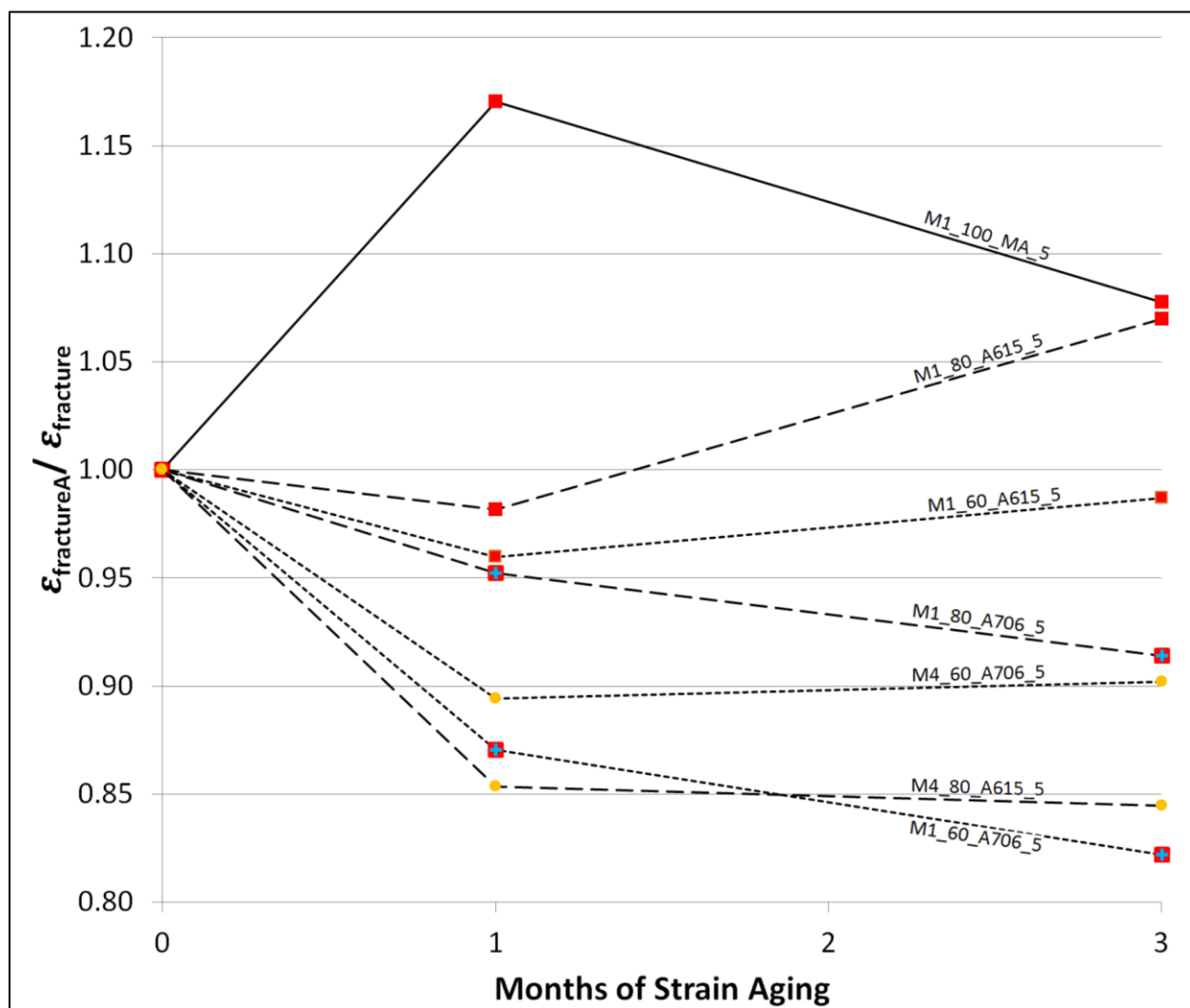


Figure 17: Normalized $\epsilon_{\text{fractureA}}$ vs strain aging duration

4.2.4.2 Effects of Vanadium on Strain Aging

In Figure 18 and Figure 19, clear relationships can be seen between Vanadium concentrations and the effects of strain aging. The apparent yield strength of strain-aged bars decreases with increasing concentrations of Vanadium, up to a concentration of about 0.08% (Figure 18). Beyond that concentration, $(\Delta\sigma/f_y)$ appears to level off at 0.03 regardless of the concentration of Vanadium. Changes in fracture elongation also appear to vary with the concentration of Vanadium with strain-aged bars becoming more ductile with higher Vanadium concentrations. An increase in ductility post-aging was observed at relatively high concentrations of Vanadium (in excess of 0.35%). Only two data points are available however with those high concentrations.

Based on these observations, the following relations were developed between Vanadium concentrations in percentages (%V) and $(\Delta\sigma/f_y)$ or $(\epsilon_{\text{fractureA}}/\epsilon_{\text{fracture}})$.

$$\Delta\sigma/f_y = -1.26 * (\%V) + 0.13 \quad 0.00 \leq \%V \leq 0.08 \quad (\text{Equation 4.2.1})$$

$$\Delta\sigma/f_y = 0.03 \quad 0.08 < \%V \quad (\text{Equation 4.2.2})$$

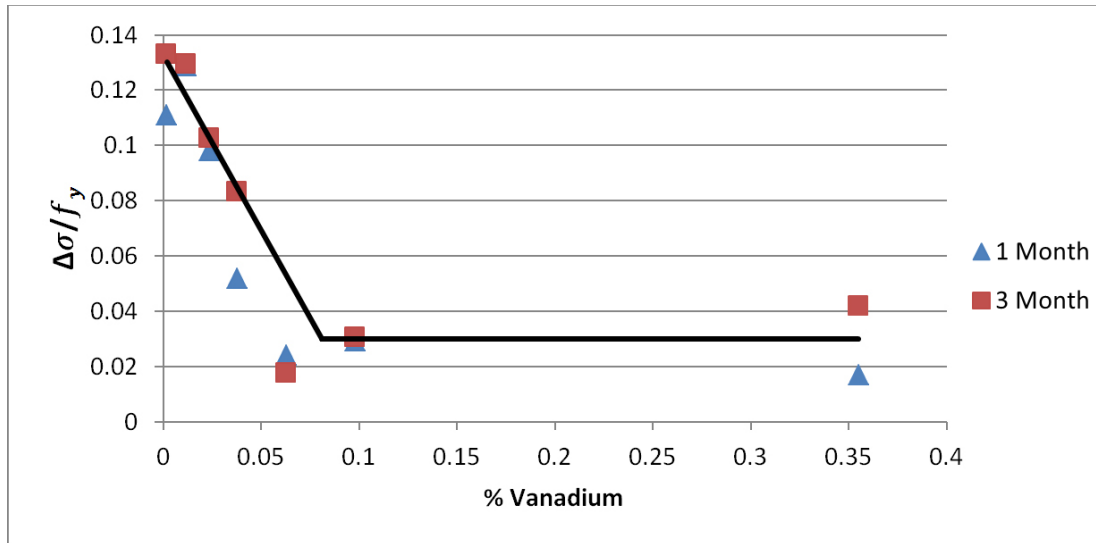


Figure 18: $\Delta\sigma/f_y$ vs %Vanadium

$$\varepsilon_{fractureA}/\varepsilon_{fracture} = 0.9 * (\%V) + 0.9 \quad 0.00 \leq \%V \leq 0.26 \quad (\text{Equation 4.2.3})$$

$$\varepsilon_{fractureA}/\varepsilon_{fracture} = 1.12 \quad 0.26 < \%V \quad (\text{Equation 4.2.4})$$

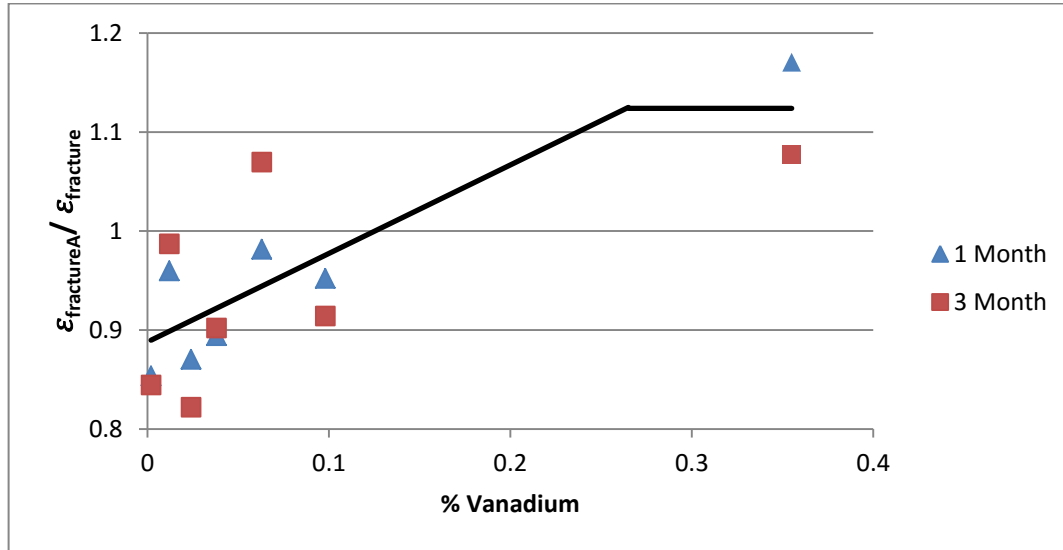


Figure 19: Normalized $\varepsilon_{fractureA}$ vs % Vanadium

4.3 Bend/re-bend Tests

A total of 60 bend/re-bend tests were performed. The following performance measures were used to compare the performance between the various bar types and grades:

- a) Fracture location (in 90° bend, 45° bend or in straight regions)
- b) Remaining bend angle at fracture during re-bending (θ_b)
- c) Axial strain in bar at fracture normalized by uniform elongation strain ($\epsilon_b / \epsilon_{un}$)
- d) Axial stress in bar at fracture normalized by yield strength (f_{ub} / f_y)
- e) Axial stress in bar at fracture normalized by tensile strength (f_{ub} / f_t)

Figure 20 shows a fracture in the 90 degree bend. The remaining bend angle during re-bending was measured using the targets bracketing the 90 degree bend. Two slopes were calculated from the two straight regions adjacent to the 90 degree bend. Using the relationship between the two slopes, the remaining bend angle was calculated, which starts around 90 degrees and goes to almost zero when a bar is fully straightened. Axial strain in bars was obtained by averaging the strains between the targets furthest away from each other in the top and bottom straight regions as indicated by the arrows in Figure 20. Axial stress in bars was obtained by dividing the load reading of the test machine by the nominal area of the bars. The average stresses for all coupons of the same bar type are presented in Table 5 and Table 6.



Figure 20: Photograph of a bar that fractured in the 90° bend after limited straightening

4.3.1 Typical Stress vs. Strain Relations in Bend/re-bend Tests

The axial stress-strain relations for three re-bend tests in which fracture occurred prior to full bend straightening are shown in Figure 21. The axial stress-strain relations for bars in which fracture occurred after straightening and after significant inelastic straining occurred in the straight regions are shown in Figure 22. As can be seen in Figures 21 and 22, the initial portion of the stress strain relations measured are not linear. This is attributed to the bending moments that develop due to second order effects during re-bending in the top and bottom straight portions of the specimens where the strains were measured (Figure 20). These moments die out once the bars straighten.

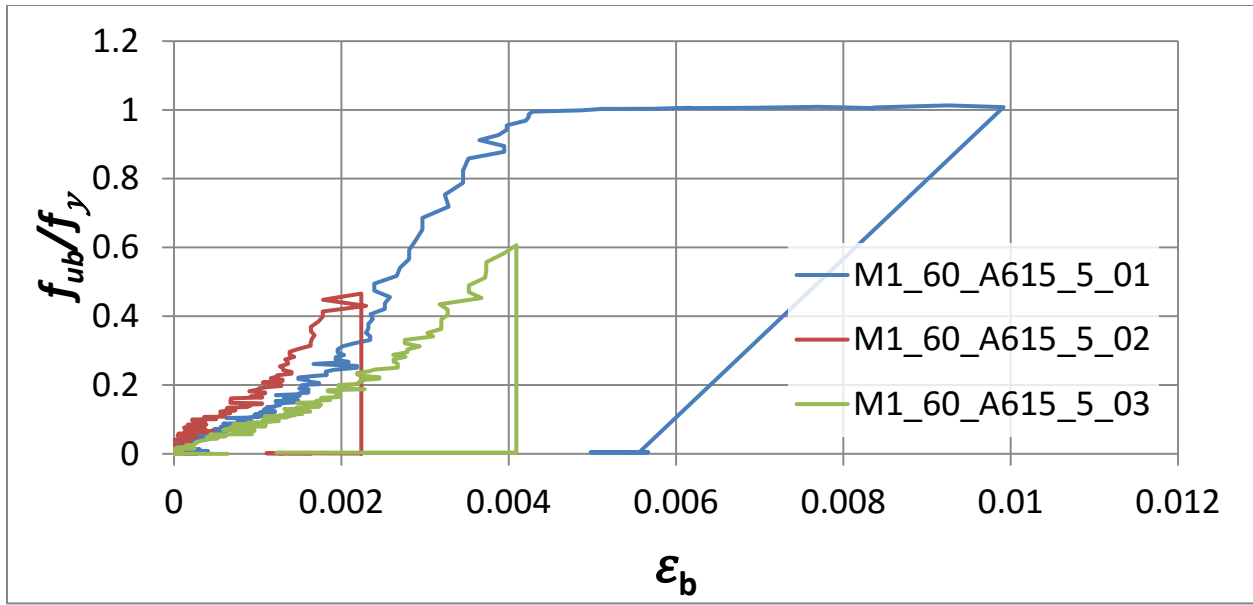


Figure 21: Typical stress vs strain relations for non-straightened bars

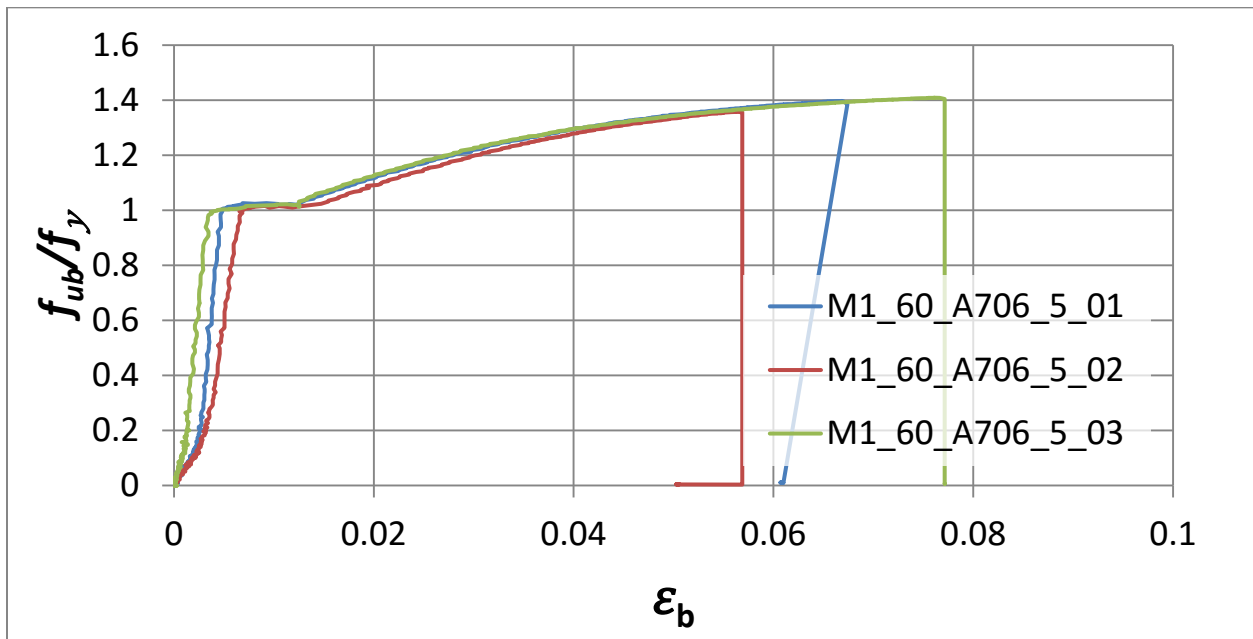


Figure 22: Typical stress vs strain relations for straightened bars exhibiting high ductility during re-bending

4.3.1 Typical Stress vs. Remaining Bend Angle Relations in Bend/re-bend Tests

Typical bar stress versus remaining bend angle relations are plotted in Figure 23 for bars that fractured prior to full straightening. Typical bar stress versus remaining bend angle relations are plotted in Figure 24 for bars that fractured after full straightening. In the figures, 90 degrees denotes the initial bend angle and 0 degrees denotes that a bar-bend has been fully straightened. Tests highlighted in Figure 23 and Figure 24 are those highlighted in Figure 21 and Figure 22, respectively.

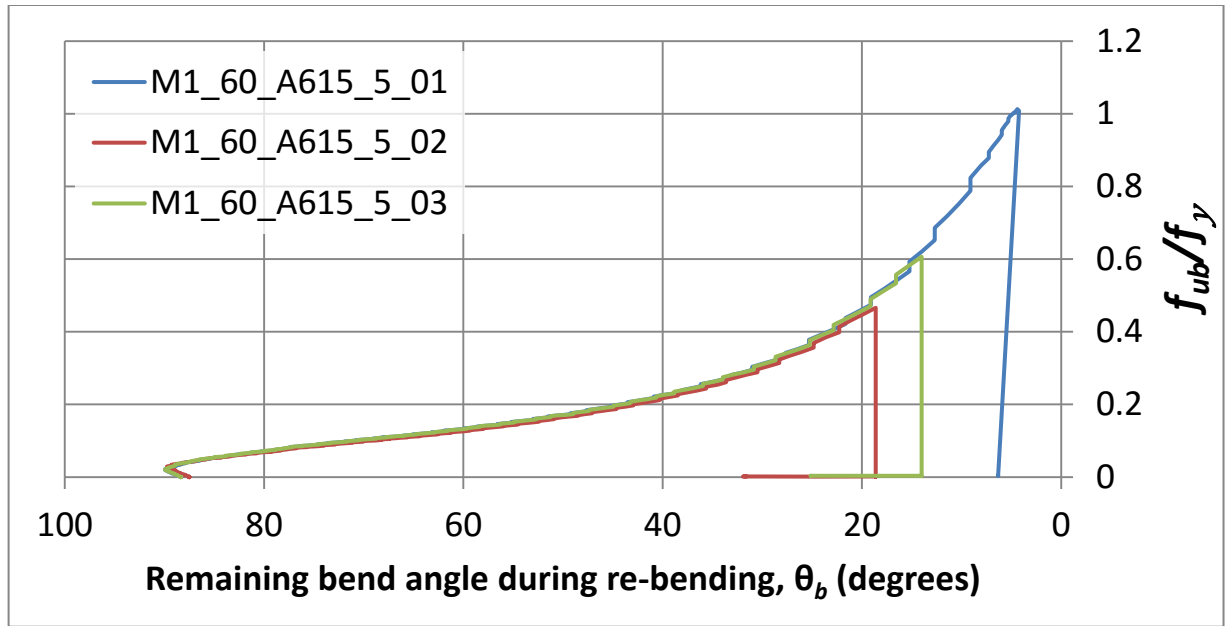


Figure 23: Typical stress vs. remaining bend angle relations for non-straightened bars

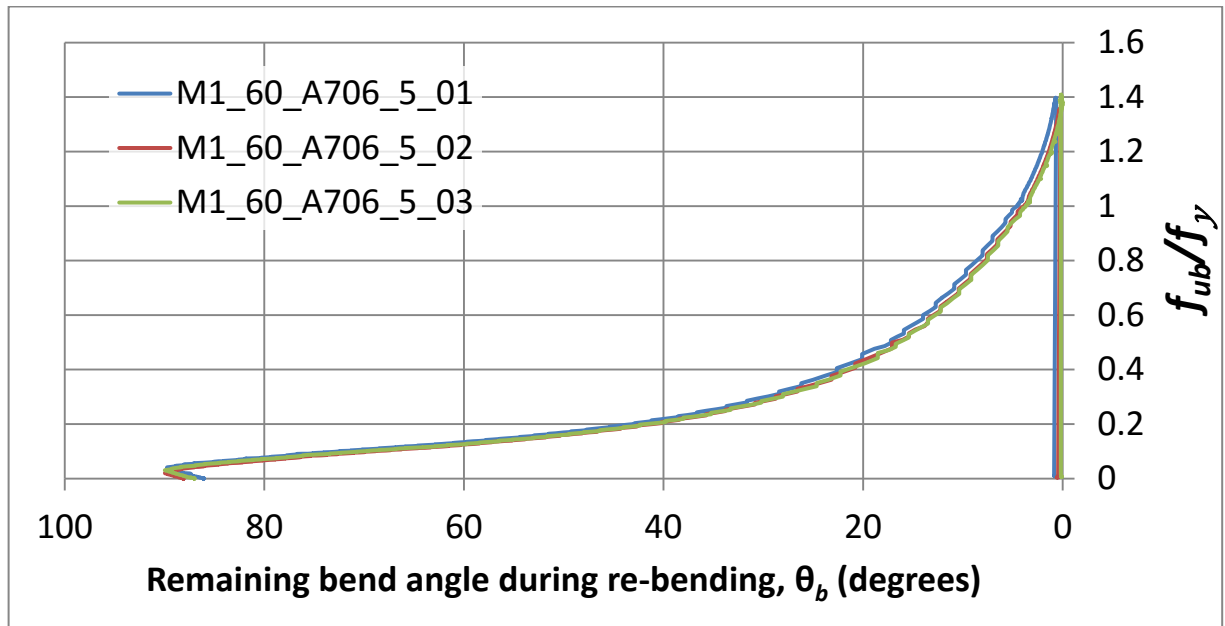


Figure 24: Typical stress vs. remaining bend angle relations for straightened bars

4.3.2 Summary of Results for Bend/re-bend Tests

Table 5 and Table 6 summarize values of the bend/re-bend test performance measures, averaged over at least three tests per bar type. Table 5 contains results from #8 and #11 bars, and Table 6 contains results from #5 bars.

The bend/re-bend tests subjected bar bends to harsher stress and strain histories than they typically encounter in concrete structures. As such, defining critical performance measure values that delineate deficient in-situ bend performance is not straightforward. However, selecting values of performance measures above which bend performance can be deemed adequate can be done conservatively. In this study, bars with values of the normalized stress at fracture (f_{ub}/f_y) exceeding 1.0 are deemed to have adequate bend performance, as these bars have reached their design strength prior to fracture. Likewise, bars with values of the normalized fracture strain ($\epsilon_{sb} / \epsilon_{un}$) exceeding 0.2 are deemed to have adequate bend performance, as these bars typically straighten fully, reach stresses in excess of yield, and strain to at least 20% of their uniform elongation prior to fracture.

Table 5: Bend/re-bend Results for #8 and #11 Bars

Bar Size (Bend Diameter)	Manf.	Grade	Specimen Name	Yield Stress f_y (ksi)	Tensile Stress f_t (ksi)	Uniform Strain ϵ_{un} (%)	Normalized Strain at Fracture (ϵ_b/ϵ_{un})	Normalized Stress at Fracture (f_{ub}/f_y)	Angle at Fracture θ_b (degrees)	Fracture Location			$\theta_b < 5$ degrees
										Straight Regions	45	90	
#11 (8d _b)	M1	A706 60	M1_60_A706_11	64.3	93.2	12.5%	1.05	1.47	0	1/3	1/3	1/3	3/3
		A706 80	M1_80_A706_11	81.7	111.2	10.3%	0.96	1.38	0	3/3			3/3
		MA 100	M1_100_MA_11	110.4	139.6	8.8%	0.52	0.83	28	2/3		1/3	2/3
		A615 60	M1_60_A615_11	63.2	104.0	11.2%	1.05	1.64	0	3/3			3/3
		A615 80	M1_80_A615_11	80.5	121.1	9.1%	0.77	1.46	1	2/3	1/3		3/3
	M2	A1035 100	M2_100_A1035_11	125.0	162.1	4.9%	0.47	1.23	1		3/3		3/3
	M3	A706 80	M3_80_A706_11	83.1	109.7	9.1%	0.86	1.32	0			3/3	3/3
	M4	A615 60	M4_60_A615_11	63.6	90.7	12.1%	1.11	1.42	0	3/3			3/3
		A706 60	M4_60_A706_11	67.5	95.8	11.5%	1.36	1.42	0	3/3			3/3
#8 (6d _b)	M1	A706 60	M1_60_A706_8	60.5	90.0	11.5%	1.01	1.50	0	3/3			3/3
		A706 80	M1_80_A706_8	84.4	114.1	9.8%	0.92	1.35	0	3/3			3/3
		A706 100	M1_100_MA_8	99.0	125.2	8.9%	1.11	1.28	0	3/3			3/3
		A615 60	M1_60_A615_8	63.7	101.3	10.7%	1.12	1.61	0	3/3			3/3
		A615 80	M1_80_A615_8	84.4	123.5	9.2%	0.73	1.45	0			3/3	3/3
	M2	A1035 100	M2_100_A1035_8	131.4	164.3	5.2%	0.43	1.17	1		3/3		3/3
	M3	A706 80	M3_80_A706_8	81.6	104.0	8.9%	0.99	1.28	0			3/3	3/3
		QT 100	M3_100_QT_8	98.7	126.0	7.2%	0.38	1.16	1		1/3	2/3	3/3
	M4	A615 60	M4_60_A615_8	68.1	95.8	12.0%	1.06	1.41	0	2/3		1/3	3/3
		A706 60	M4_60_A706_8	66.7	90.9	12.3%	1.18	1.36	0	1/3		2/3	3/3

Table 6: Bend/re-bend Results for #5 Bars

Bar Size (Bend Diameter)	Manf.	Grade	Specimen Name	Yield Stress f_y (ksi)	Tensile Stress f_t (ksi)	Uniform Strain ϵ_{un} (%)	Normalized Strain at Fracture (ϵ_b/ϵ_{un})	Normalized Stress at Fracture (f_{ub}/f_y)	Angle at Fracture θ_b (degrees)	Fracture Location			$\theta_b < 5$ degrees
										Straight Regions	45	90	
#5 (4d _b)	M1	A706 60	M1_60_A706_5	65.7	93.9	10.5%	0.64	1.39	0			3/3	3/3
		A706 80	M1_80_A706_5	86.5	115.2	9.5%	0.03	0.28	26			3/3	0/3
		A706 100	M1_100_MA_5	113.0	135.1	8.2%	0.01	0.15	52			3/3	0/3
		A615 60	M1_60_A615_5	63.0	97.9	11.2%	0.05	0.70	12			3/3	1/3
		A615 80	M1_80_A615_5	81.8	112.9	9.9%	0.03	0.25	31			3/3	0/3
	M2	A1035 100	M2_100_A1035_5	125.6	163.6	5.4%	0.11	0.51	26			3/3	1/3
	M3	A706 80	M3_80_A706_5	83.3	102.7	8.8%	0.24	1.06	1			3/3	3/3
		QT 100	M3_80_QT_5	90.0	129.7	6.9%	0.02	0.12	61			3/3	0/3
	M4	A706 60	M4_60_A706_5	66.0	90.8	10.3%	0.51	1.28	0			3/3	3/3
		A615 60	M4_60_A615_5	80.2	102.9	12.1%	0.44	1.22	0			3/3	3/3
		MA 100	M4_100_MA_5	106.1	125.6	7.6%	0.04	0.71	7			3/3	2/3
#5 (5d _b)	M1	A706 60	M1_60_A706_5	65.7	93.9	10.5%	1.01	1.41	0	1/3	1/3	1/3	3/3
		A706 80	M1_80_A706_5	86.5	115.2	9.5%	0.53	1.28	17			3/3	1/3
		A706 100	M1_100_MA_5	113.0	135.1	8.2%	0.36	0.70	12	1/3		2/3	1/3
		A615 60	M1_60_A615_5	63.0	97.9	11.2%	0.62	1.49	0	1/3		2/3	3/3
		A615 80	M1_80_A615_5	81.8	112.9	9.9%	0.02	0.52	0			3/3	3/3
	M2	A1035 100	M2_100_A1035_5	125.6	163.6	5.4%	0.24	0.86	17		1/3	2/3	2/3
	M3	A706 80	M3_80_A706_5	83.3	102.7	8.8%	0.62	1.21	0			3/3	3/3
		QT 100	M3_80_QT_5	90.0	129.7	6.9%	0.02	0.23	36			3/3	0/3
	M4	A706 60	M4_60_A706_5	66.0	90.8	10.3%	1.10	1.35	0			3/3	3/3
		A615 60	M4_60_A615_5	80.2	102.9	12.1%	0.67	1.27	0			3/3	3/3
		MA 100	M4_100_MA_5	106.1	125.6	7.6%	0.19	1.00	2			3/3	3/3
#5 (6d _b)	M1	A706 100	M1_100_MA_5	113.0	135.1	8.2%	1.01	1.18	0	2/3		1/3	3/3
	M2	A1035 100	M2_100_A1035_5	125.6	163.6	5.4%	0.47	1.25	0			3/3	3/3
	M3	A706 80	M3_80_A706_5	83.3	102.7	8.8%	0.73	1.21	0			3/3	3/3
		QT 100	M3_80_QT_5	90.0	129.7	6.9%	0.04	0.44	21			3/3	0/3
	M4	MA 100	M4_100_MA_5	106.1	125.6	7.6%	0.32	1.08	0		2/3	1/3	3/3

4.3.3 Effects of Bar Yield Strength on Re-bend Performance

As can be seen in Figure 25 to 28 the performance of bar bends generally decreased as the measured yield strength of the bars increased. This trend holds across all bar sizes.

A negative correlation can be observed between (f_{ub}/f_y) during re-bending and the measured bar yield strength (Figure 25). For #8 bar specimens with an inside bend diameter of $6d_b$ (or #8 ($6d_b$)) and for #11 ($8d_b$) specimens, test data exhibited relatively low variability about the observed trends highlighted by the linear regression lines in Figure 25. However, for #5 ($4d_b$) specimens, test data exhibited relatively large variability about the observed trend. The #8 ($6d_b$) and #11 ($8d_b$) specimens consistently failed at stresses well above the yield strength and closer to the tensile strengths of the bars at all bar strength levels; with the exception of the M1_Gr100_MA_11 specimen. To decouple the trend of lower tensile-to-yield-strength ratios with increasing yield strength from bend performance, the stresses at fracture normalized by the tensile strength of the bars (f_{ub}/f_t) are plotted versus the measured yield strength of the bars in Figure 26. As can be seen in the figure, the negative correlation can be observed for #5 bars as yield strength increases. #8 and #11 bars seem to reach more than 95% of f_t with the exception of M3_Gr100_QT_8 and M1_Gr100_MA_11. This suggests that the tensile-to-yield-strength ratio might falsely reflect the performance measure (f_{ub}/f_y) as f_y increases.

A negative correlation between the yield strength and the normalized strain at fracture can also be observed in Figure 27. Most #8 bars ($6d_b$) and #11 ($8d_b$) bars, regardless of yield strength, straightened almost fully during re-bending (Figure 28). For #5 ($4d_b$) bars, however, there is a clear increase in the remaining bend angle at fracture as the bar yield strength increased.

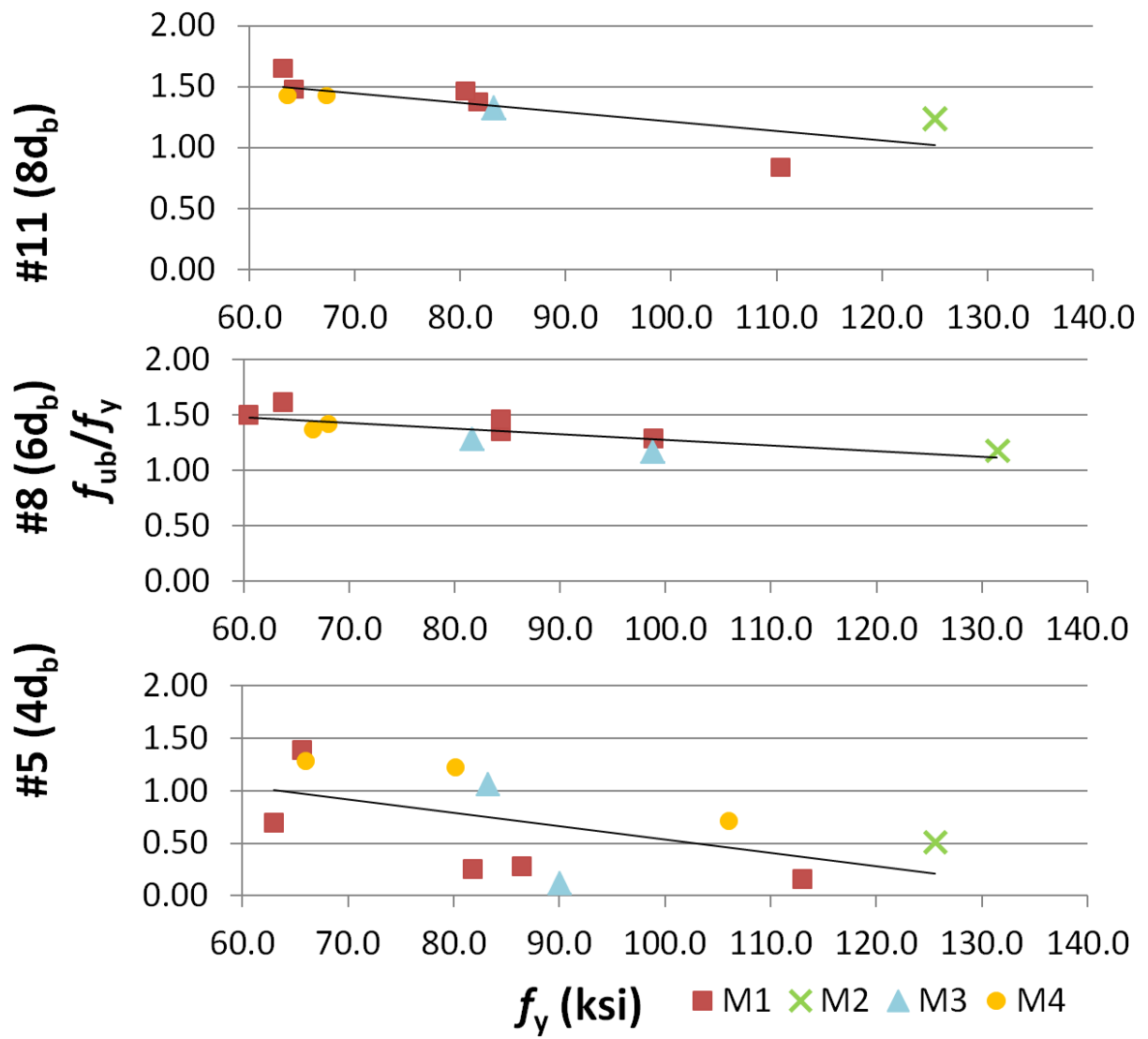


Figure 25: Normalized re-bend stress at fracture vs measured bar yield strength

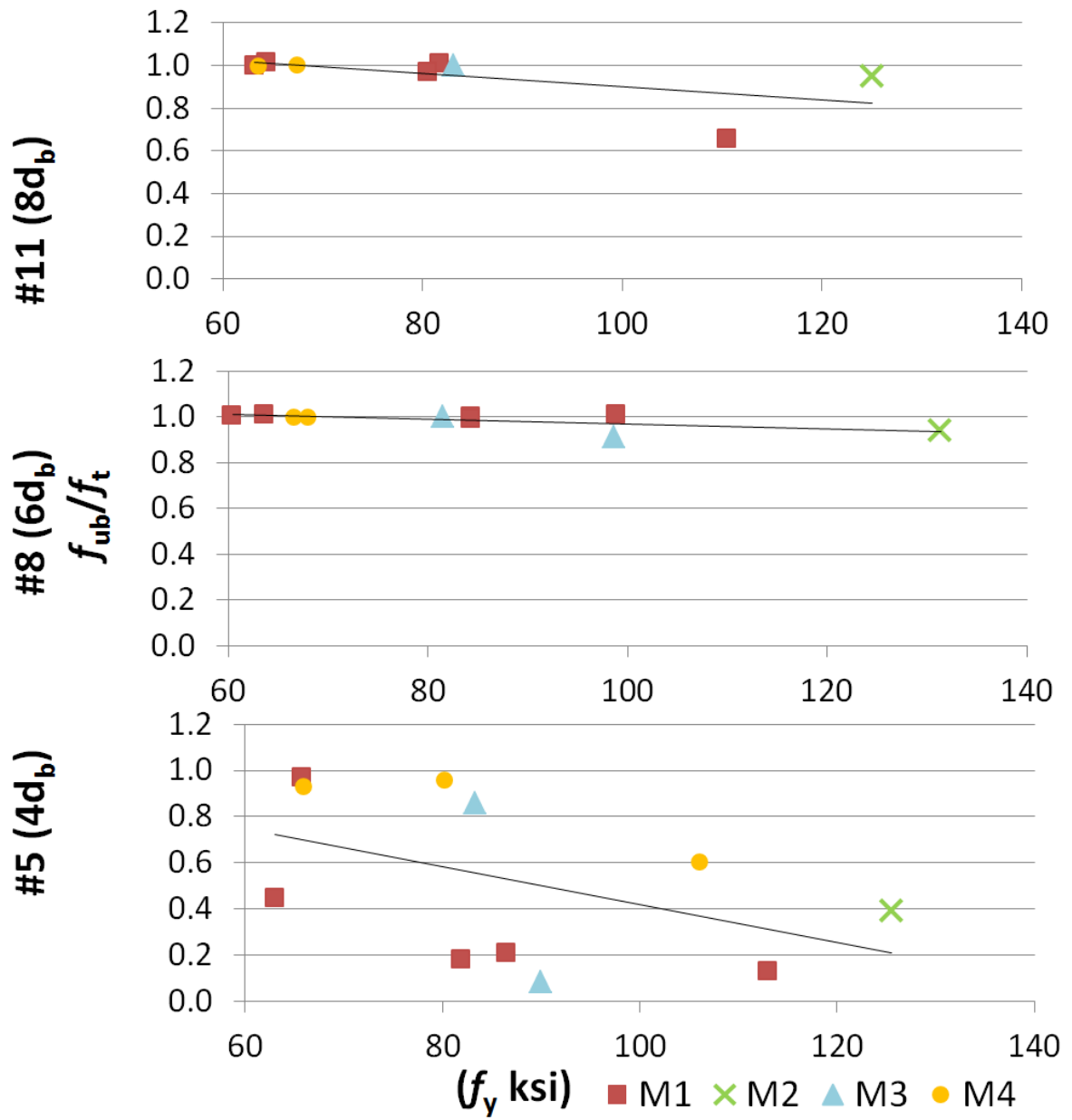


Figure 26: Normalized re-bend stress at fracture vs measured bar yield strength

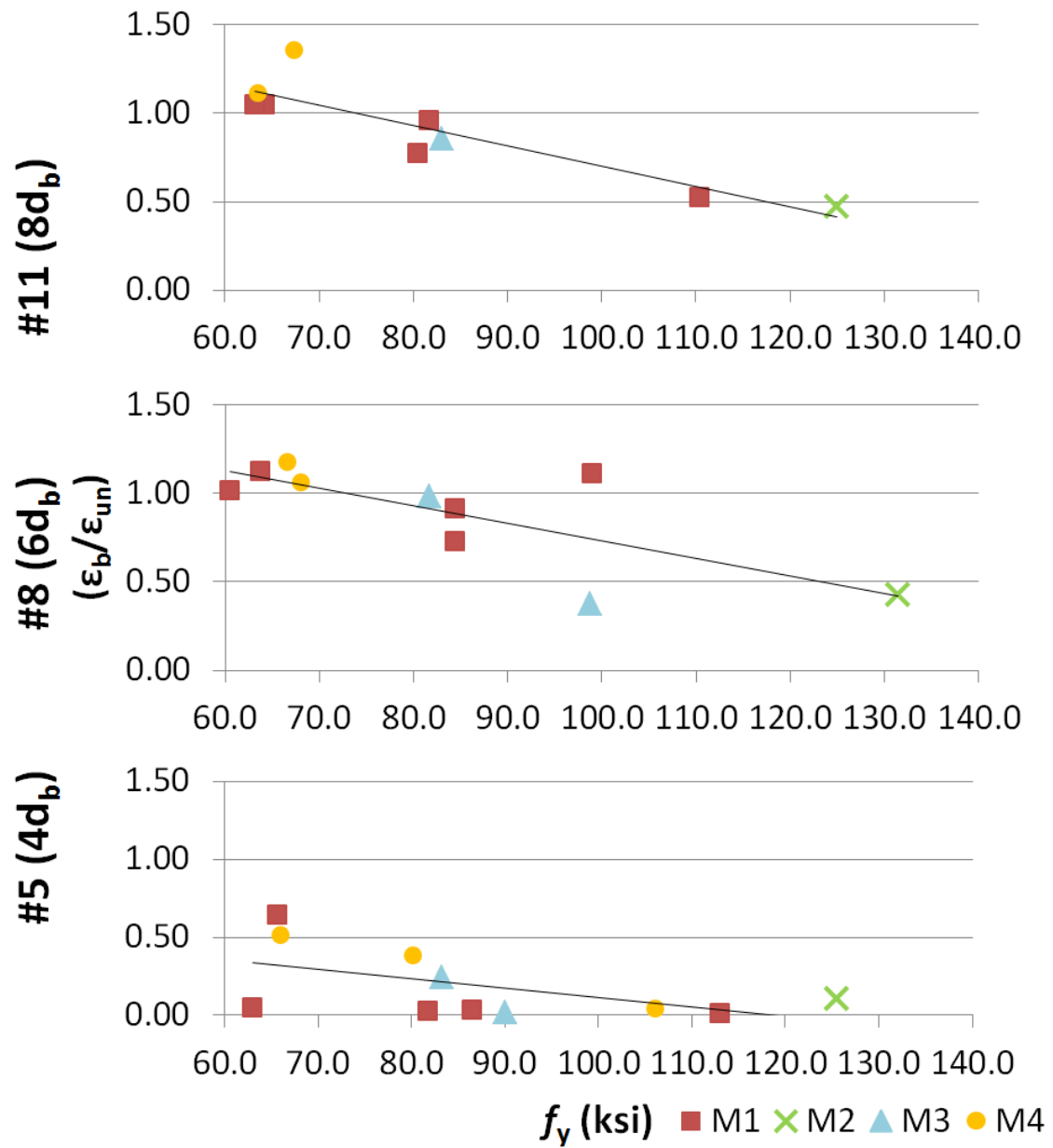


Figure 27: Normalized re-bend strain at fracture vs measured bar yield strength

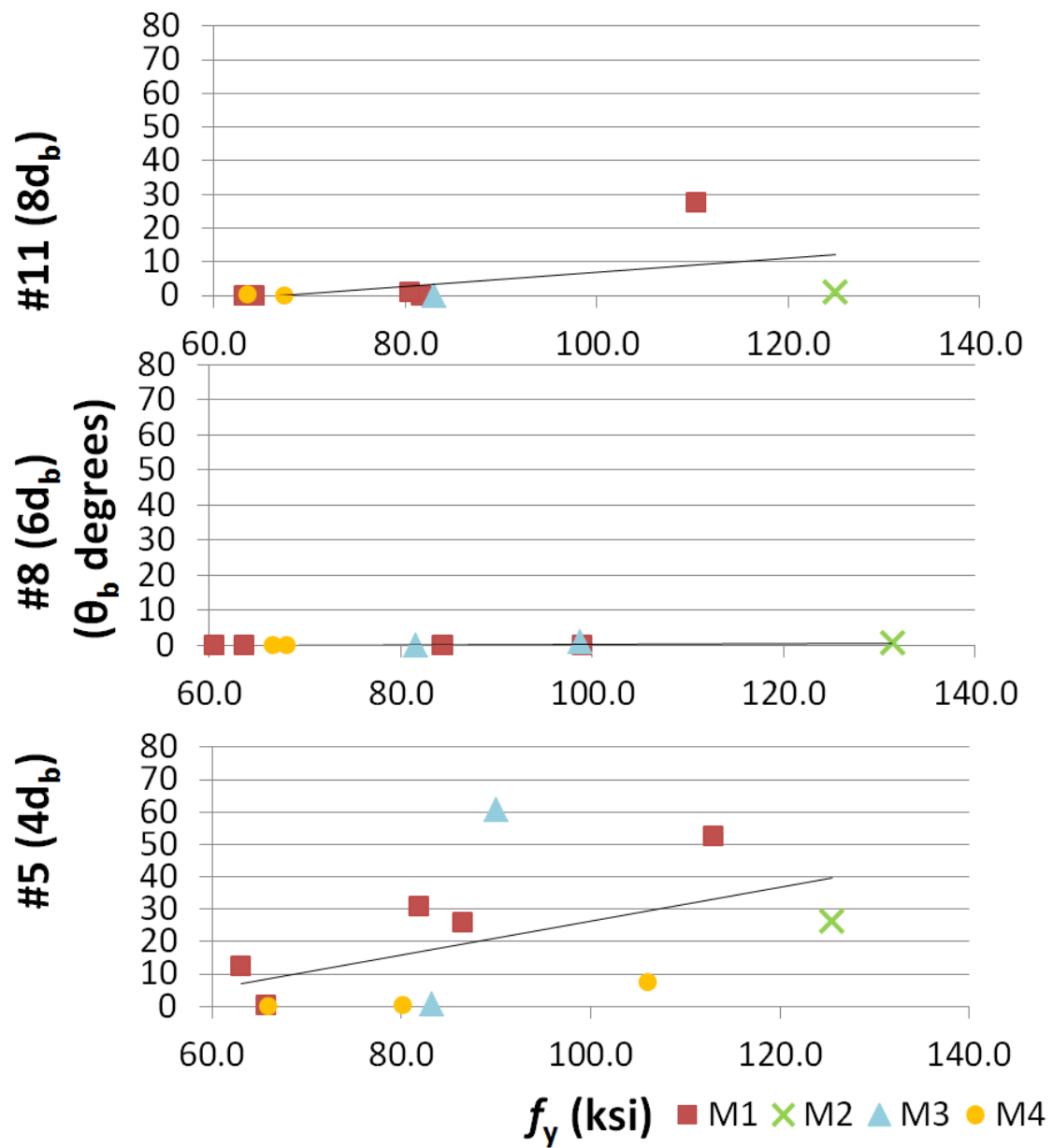


Figure 28: Remaining bend angle at fracture vs measured bar yield strength

4.3.4 Effects of Bar Size on Re-bend Performance

The relationship between (f_{ub}/f_y) and bar yield strength is compared for different bar sizes in Figure 29. As can be seen in the figure, #11 ($8d_b$) and #8 ($6d_b$) bars reached significantly larger stresses at fracture than #5 ($4d_b$) bars. In fact, except for one bar type, all #11 and #8 bars reached stresses in excess of their yield strength during re-bending. However, as the bend diameter of #5 bars was increased from $4d_b$, the bend performance of #5 bars improved and became comparable to that of the larger bars when the inside bend diameter of the #5 bars was $6d_b$ (same as that of #8 bars). Test data therefore suggest a limited influence of bar size on the bar stress at fracture during re-bending, but a significant influence of bend inside diameter on stress at fracture. Similar trends can be observed between bar size and the normalized strain at fracture Figure 30.

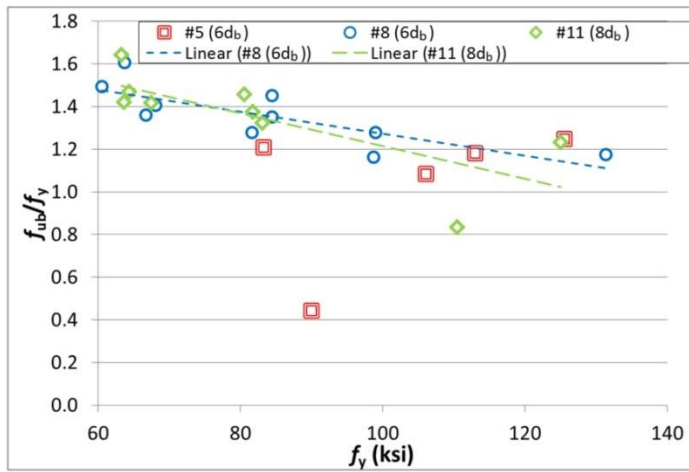
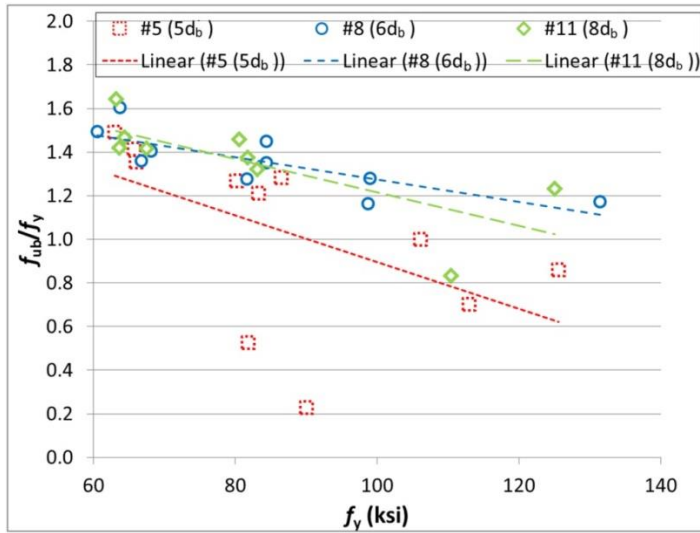
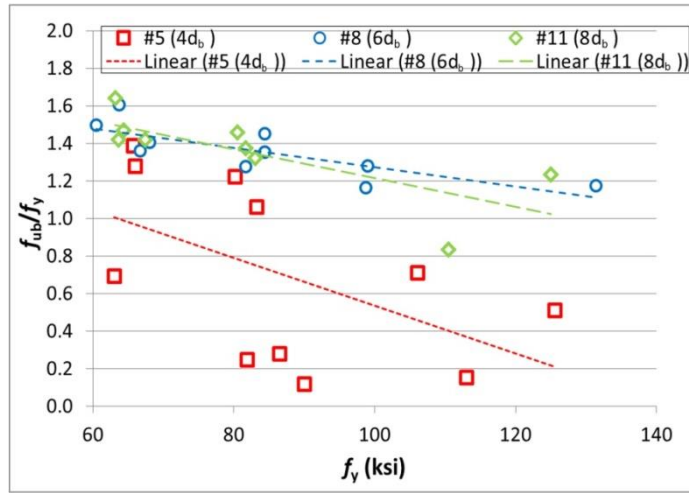


Figure 29: Comparison of normalized stress at fracture vs measured yield strength for various bar sizes

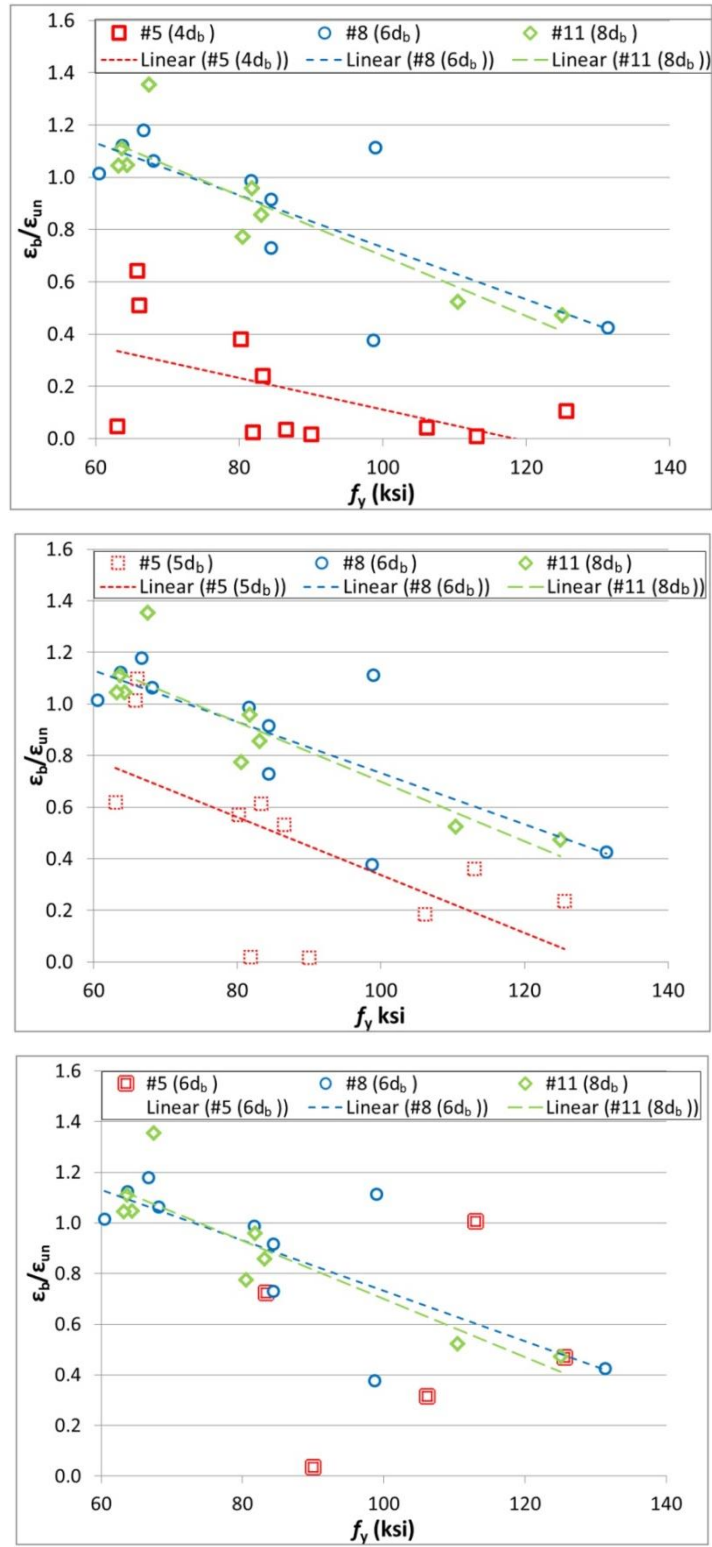


Figure 30: Comparison of normalized strain at fracture vs measured yield strength for various bar sizes

Given the relatively large scatter in the performance of #5 bars compared with that of the larger bars, the possible effects of the manufacturing processes on the performance of #5 bars are explored in Figure 31. The #5 bars from Manufacturer 1 bent at $4d_b$, with the exception of M1_Gr60_A706 bars, fractured at stresses below the linear regression trend line for all #5 ($4d_b$) bars. However, the fractures stresses of bars from Manufacturer 1 bent at $5d_b$ were distributed above and below the trend line, but exhibited high variability. Bars produced by Manufacturer 3 showed the high variability as well, with some bars failing at significantly higher stresses than the trend lines and others at significantly lower stresses. Bars from Manufacturers 2 and 4 were consistently above the trend lines.

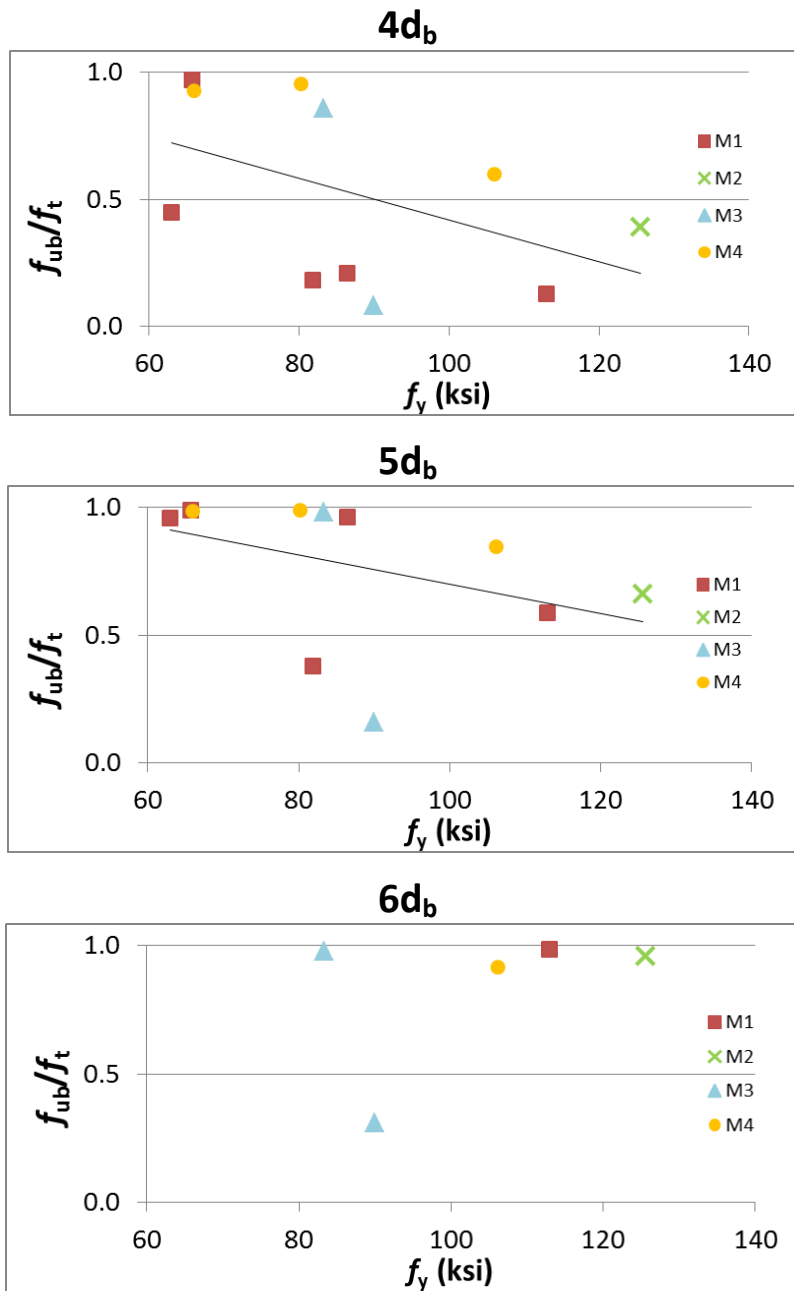


Figure 31: Re-bend stress at fracture normalized by tensile strength vs measured yield strength for #5 bars

4.3.6 Effects of Bend Diameter on Re-bend Performance

The #11 ($8d_b$) and #8 ($6d_b$) bar specimens performed reasonably well across all grades. However, the #5 ($4d_b$) bars did not. For this reason, additional #5 bars were bent at inside bend diameters of $5d_b$ and $6d_b$.

The effects of increasing bend diameters on (f_{ub}/f_y) can be seen in Figure 32. In the figure, lines with short dashes denote bars with yield strengths around 60 ksi (grade 60), lines with longer dashes denote bars with yield strengths around 80 ksi (grade 80), and solid lines denote bars with yield strengths around or exceeding 100 ksi (grade 100). As can be seen in

Figure 32, the stress at fracture of grade 60 bars is not affected significantly when increasing bend diameters from $4d_b$ to $5d_b$; with the exception of M1_60_A615 specimens. This is because these bars are failing at stresses close to their tensile strength at both bend diameters (Figure 31). On the other hand, the higher grades 80 and 100, fractured at stresses significantly below their tensile strength at a bend diameter of $4d_b$. This may be the reason why bars of these grades experienced significantly higher stresses at fracture with larger bend diameters.

While most grade 60 bars fractured at stresses above yield with bend diameters of $4d_b$, the majority of grade 80 and 100 bars did not. Even with a bend diameter of $5d_b$, the majority of grade 80 and 100 bars still did not reach their yield strengths prior to fracture during re-bending. It is only when the bend diameter is increased to $6d_b$ that the majority of grade 80 and 100 bars reached their yield strengths at fracture.

Considering the strain performance measure $(\epsilon_b / \epsilon_{un})$, all grades saw increases in strains at fracture with increasing bend diameters, with only a few reaching their uniform elongations during re-bending (Figure 33). Interestingly, even though grade 60 bars did not see significant increases in their stresses at fracture, they did see marked increases in their strain at fracture during re-bending.

Considering the remaining bend angle at fracture (Figure 34), all grade 60 and 80 specimens bent at $5d_b$ essentially straightened during re-bending and most did so with $4d_b$ bends. Most grade 100 bars, on the other hand, fully straightened only when they were bent at $6d_b$ and most did not straighten when bent to tighter diameters.

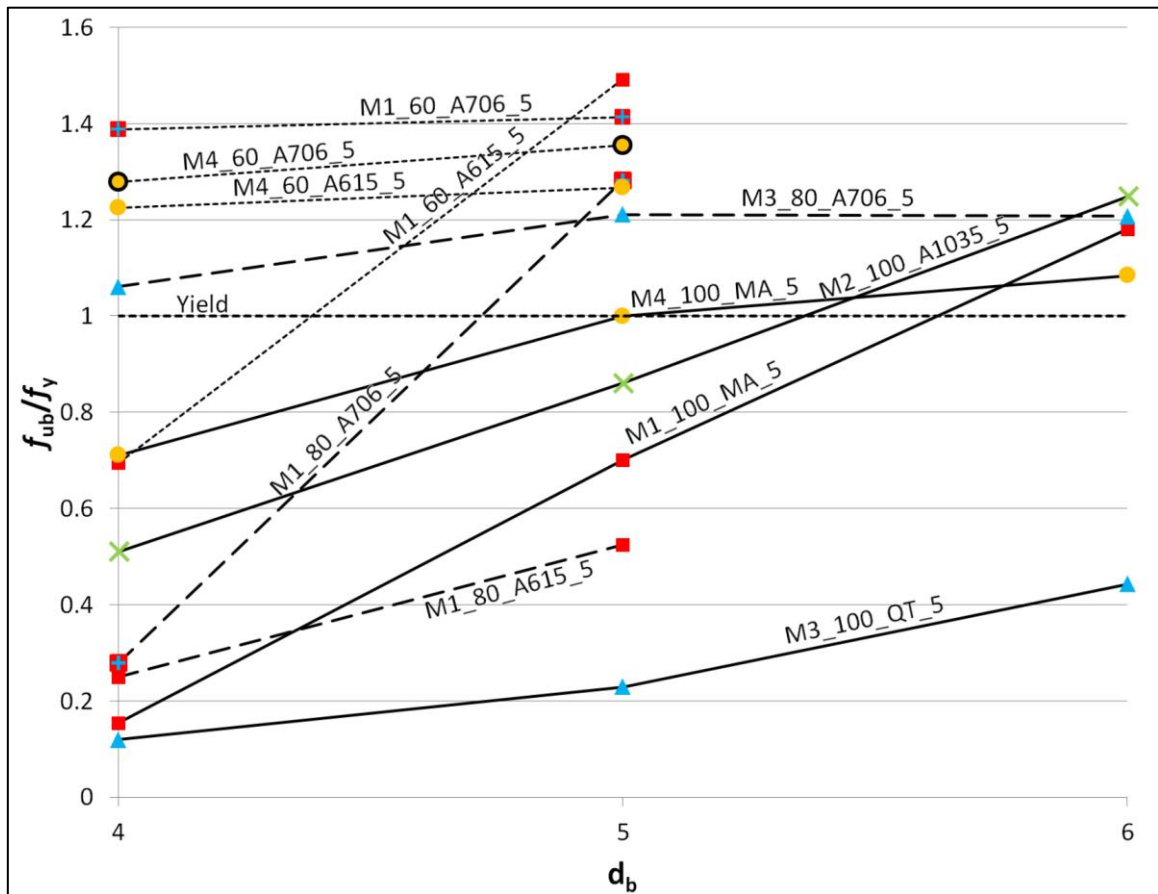


Figure 32: Normalized Re-bend Fracture Stress vs Bend Diameter

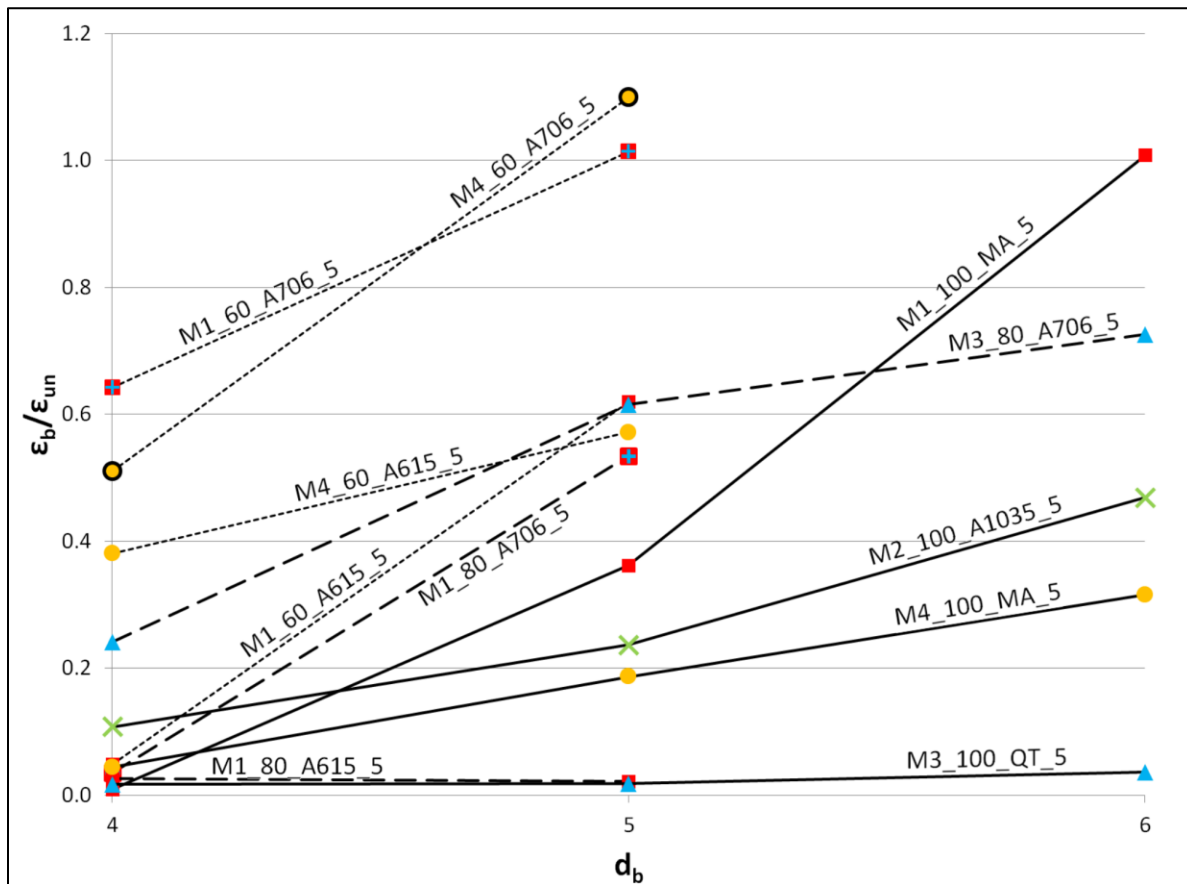


Figure 33: Normalized Re-bend Fracture Strain vs Bend Diameter

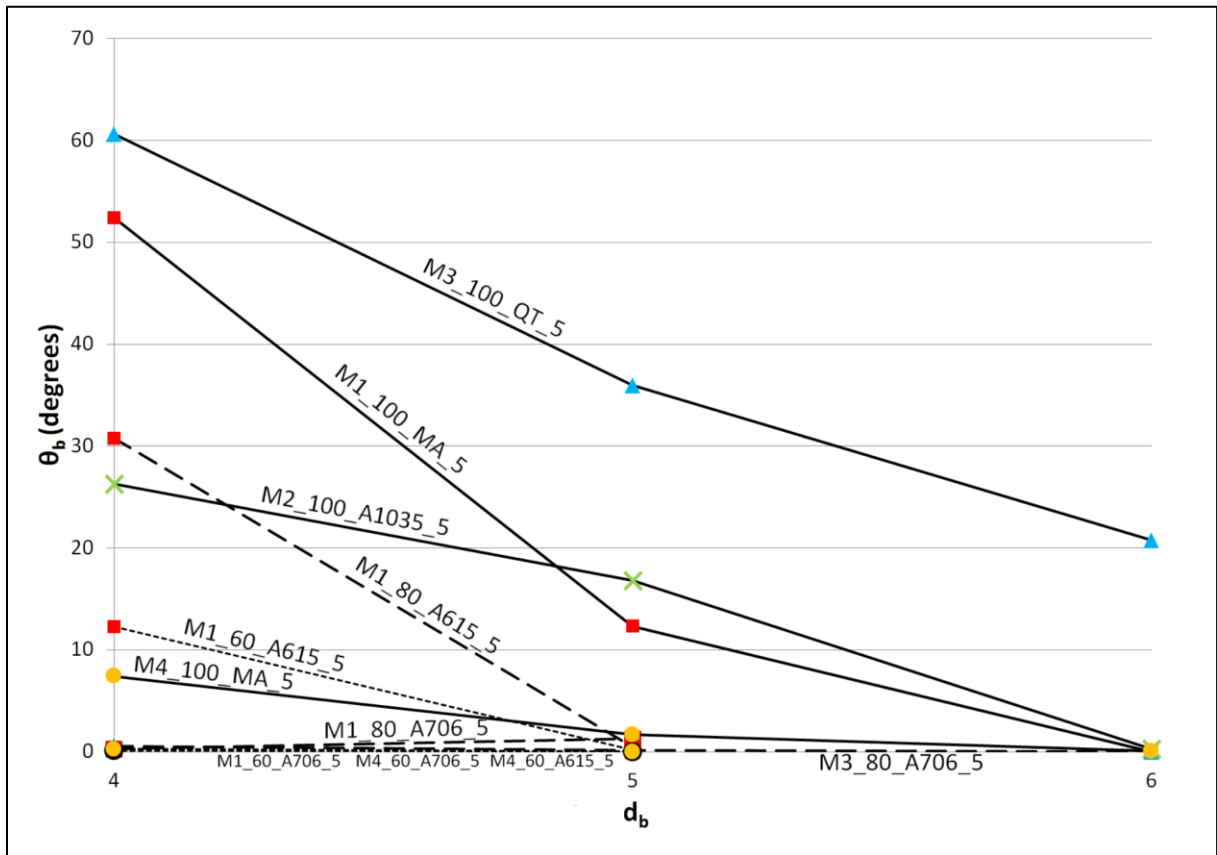


Figure 34: Normalized Re-bend Fracture Angle vs Bend Diameter

For all performance measures considered increasing the bend diameter from $4d_b$ to $6d_b$ was found to generally have a positive effect. Overall, #5 bars of all grades saw increased ductility at the bends with increasing bend diameters. Grade 80 and 100 bars saw increases in stress at fracture during re-bending with increasing bend diameters. Grade 60 bars, however did not as they reached stresses close to their tensile strength even with a bend diameter of $4d_b$.

4.3.7 Theoretical Strain in Bent Specimens

The impact of the lower ductility of HSRB on bend performance is investigated by comparing the strains induced by bending with the strain capacity of bars, namely their uniform elongation (ϵ_{un}). Since bending strains are difficult to measure, the maximum theoretical tension strain at the outer surface of the bars was estimated assuming zero elongation at the neutral axis ($\epsilon_{theoretical}$) .

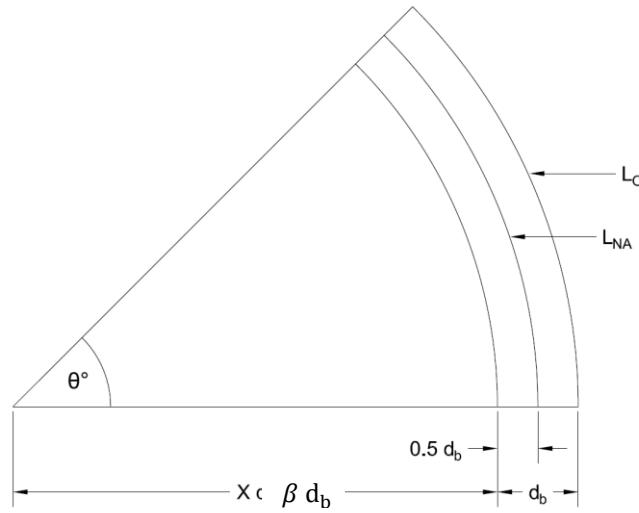


Figure 35: Max Theoretical Strain Derivation

$$\epsilon_{theoretical} = \frac{L_O}{L_{NA}} \quad (\text{Equation 4.3.1})$$

$$L_O = 2 * \pi * (\beta * d_b + d_b) * \frac{\theta}{360} \quad (\text{Equation 4.3.2})$$

$$L_{NA} = 2 * \pi * \left(\beta * d_b + \frac{1}{2} d_b \right) * \frac{\theta}{360} \quad (\text{Equation 4.3.3})$$

$$\epsilon_{theoretical} = \frac{\beta * d_b + d_b}{\beta * d_b + \frac{1}{2} d_b} - 1 = \frac{\beta + 1}{\beta + \frac{1}{2}} - 1 \quad (\text{Equation 4.3.4})$$

As can be seen in Equation 4.3.4, the theoretical strain is independent of bar size and is only a function of β , the ratio of bend diameter to bar diameter.

Using Equation 4.3.4, the maximum theoretical strain, ϵ_{mt} , experienced at the edge of the specimens using the targeted ACI 318-14 minimum bend diameters was calculated. Similarly, the maximum theoretical bend strain corresponding to the actual pin sizes used during bending, ϵ_{ma} , was calculated using Equation 4.3.4. The strains based on pin diameter rather than those based on the final bend diameter after spring back (Table 1) were considered as they were deemed to be more representative of the maximum strains experienced during bending. The calculated strain values are presented in Table 7.

Table 7: Theoretical Strains at Bends

	Bar Specimens				
	#5 (4d _b)	#5 (5d _b)	#5 (6d _b)	#8 (6d _b)	#11 (8d _b)
Pin Diameters Used (in.)	2.0	2.5	3.0	5.0	10.0
$\beta_{Target} = \text{Target Bend Diameter} / d_b$	4	5	6	6	8
$\beta_{Actual} = \text{Pin Diameter} / d_b$	3.20	4.00	4.80	5.00	7.27
Target Maximum Theoretical Strain, ϵ_{mt}	0.110	0.091	0.077	0.077	0.059
Max Theoretical Strain Based on Pin Size, ϵ_{ma}	0.135	0.111	0.094	0.091	0.064

Figure 36 compares the maximum theoretical strains associated with the bending pin diameters with the uniform strains of the bars tested. A point below the theoretical strain line suggests that the bar specimen was bent past its uniform strain. As can be seen in the figure, all but one of the #11 bars were bent to a maximum estimated strain that is considerably lower than the uniform strains of the bars. This is owing to the large ratio of bend to bar diameter used for #11 bars (β_{Actual}). Most #8 bars were also strained significantly less than uniform strains (Figure 36). All #5 bars, including grade 60 bars, were strained past their uniform strain with $\beta_{Actual} = 3.2$ (or a 4d_b target bend diameter). In addition, most #5 bars, especially higher grade bars, were strained past their uniform strain with $\beta_{Actual} = 4.0$ (or a 5d_b target bend diameter). It is only when #5 bars are bend

to a $\beta_{\text{Actual}} = 4.8$ (or a $6d_b$ target bend diameter), that grade 60 and 80 bars experience an estimated bend strain at or below their uniform strains. Grade 100 # 5 bars, however, still appeared to have been strained higher than their uniform strain values at $\beta_{\text{Actual}} = 4.8$.

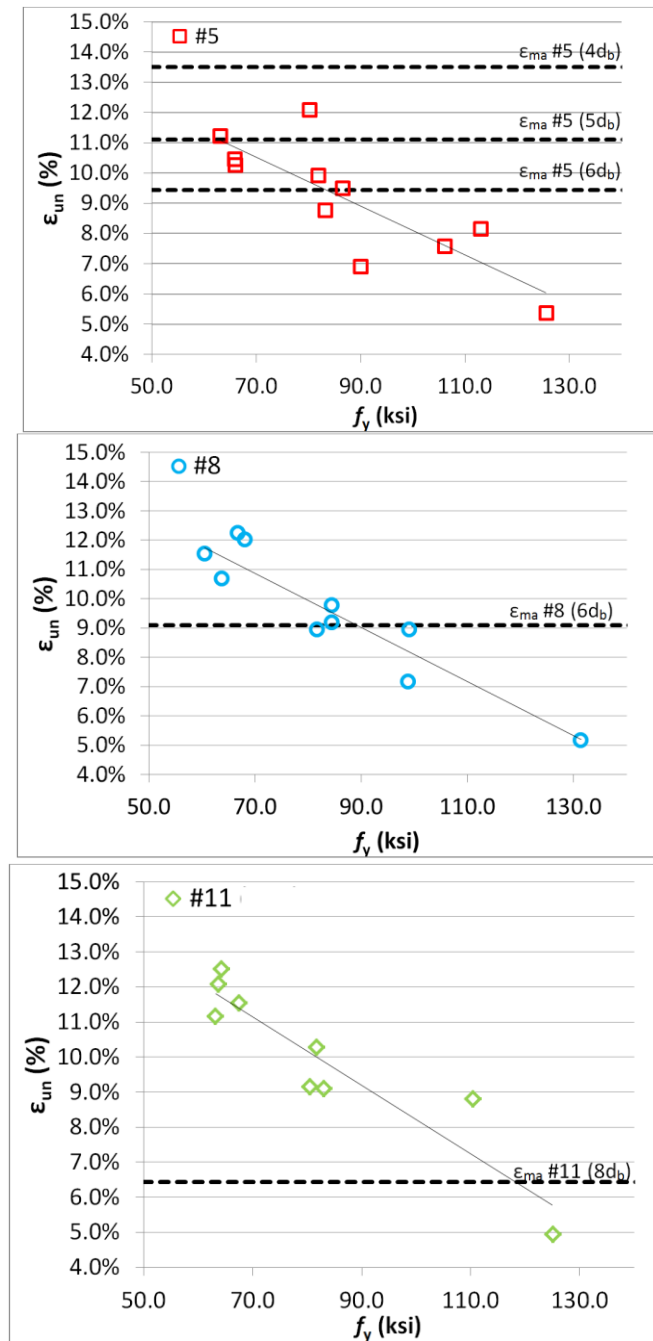


Figure 36: Bar uniform strain vs measured yield strength overlaid with the estimated maximum bend strains (ϵ_{ma})

These observed trends between strains experienced and bar uniform strains may help explain the poorer performance of #5 bars compared with the larger bars, and the poorer performance of higher grade bars that typically have lower uniform elongations than their lower grade counterparts. To explore these relations further, $(\epsilon_{ma}/\epsilon_{un})$ values were plotted versus (f_{ub}/f_y) for all bars in Figure 37, versus (f_{ub}/f_t) in Figure 38 and versus $(\epsilon_b/\epsilon_{un})$ in Figure 39. These figures indicate a clear negative correlation between the ratio of theoretical maximum strains incurred during bending and bar uniform strains $(\epsilon_{ma}/\epsilon_{un})$ for all performance measures considered. The figures therefore corroborate the hypothesis that bends in #5 and higher grade bars showed poorer performance because they were strained higher with respect to their uniform elongations.

Relations can be drawn between performance measures and the demand parameter $(\epsilon_{ma}/\epsilon_{un})$ as seen in Figure 37 to Figure 39. Given a target performance measure, these figures allow the selection of the maximum permissible bending strain to uniform strain $(\epsilon_{ma}/\epsilon_{un})$ allowable in bending. For a performance objective defined as $f_{ub}/f_y \geq 1.0$ during re-bending, Figure 37 indicates that $\epsilon_{ma}/\epsilon_{un}$ should not exceed about 1.2 during bending. With the exception of an outlying #11 data point and one #5 bar data point, we can see that all bar specimens with $\epsilon_{ma}/\epsilon_{un} \leq 1.2$ were able to develop their yield strength during re-bending (Figure 37). Moreover, with the exception of a limited number of specimens, those with an $\epsilon_{ma}/\epsilon_{un} \leq 1.2$ were also able to achieve stresses during re-bending that exceed 80% of their tensile strength (Figure 38), and strains that exceed 50% of their uniform strain capacities (Figure 39).

According to an $\epsilon_{ma}/\epsilon_{un} \leq 1.2$ criteria, the bend pin diameters used for #11 (8d_b), #8 (6d_b) and #5 (6b) bars (Table 7) result in adequate bend performance for all bar grades (Figure 36). A tighter pin diameter of four times the bar diameter, $\beta_{Actual} = 4.0$, used for #5 (5d_b) specimens, can be permitted for grade 60 and 80 bars but not grade 100 bars. Finally a $\beta_{Actual} = 3.2$ value should only be used for grade 60 bars.

Other limits on $\varepsilon_{\text{ma}}/\varepsilon_{\text{un}}$ can also be selected based on other performance objectives using Figure 37 to Figure 39.

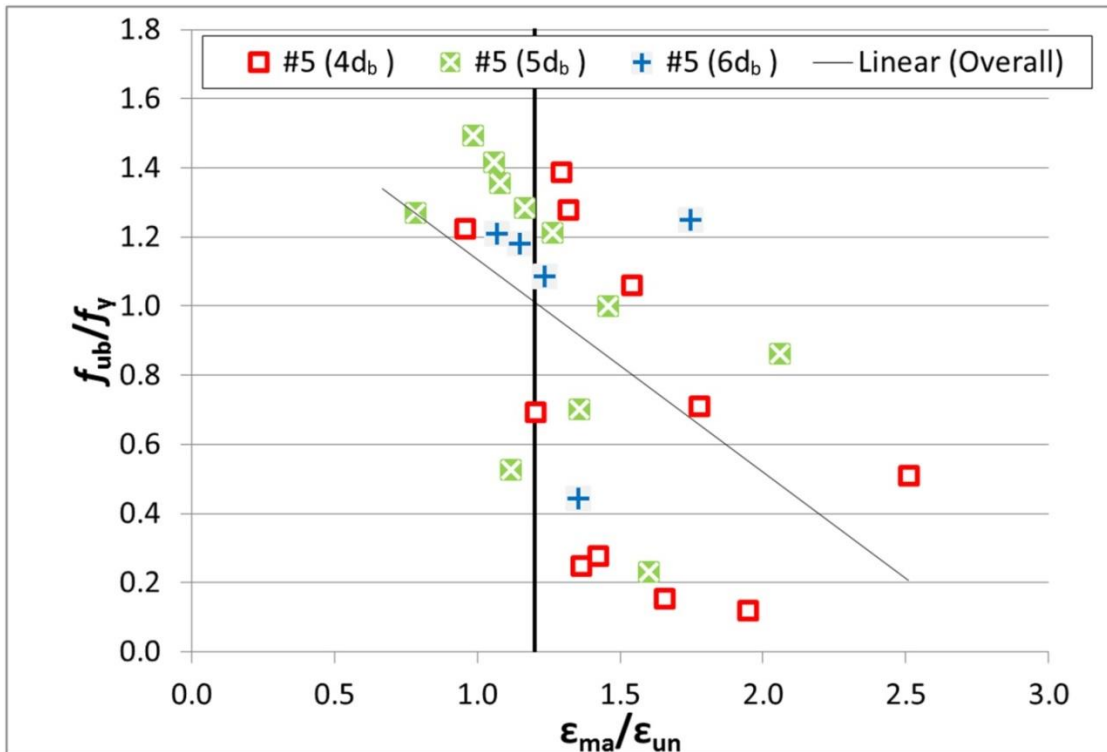
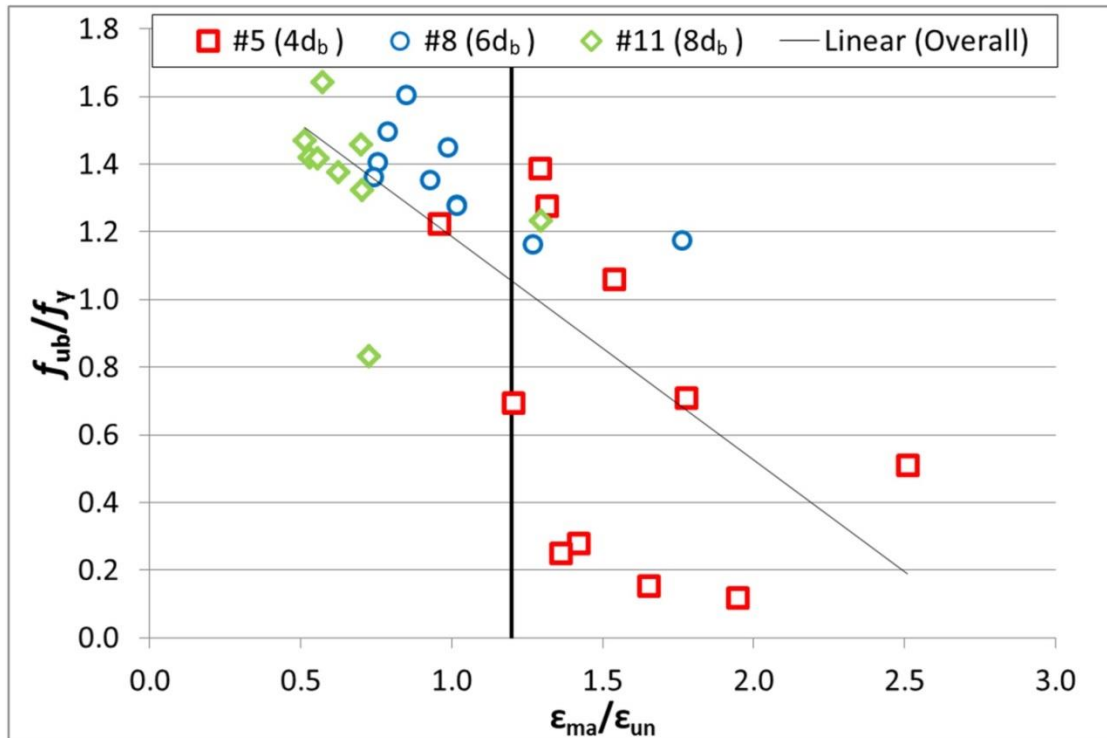


Figure 37: Normalized fracture stress vs normalized theoretical bend strain

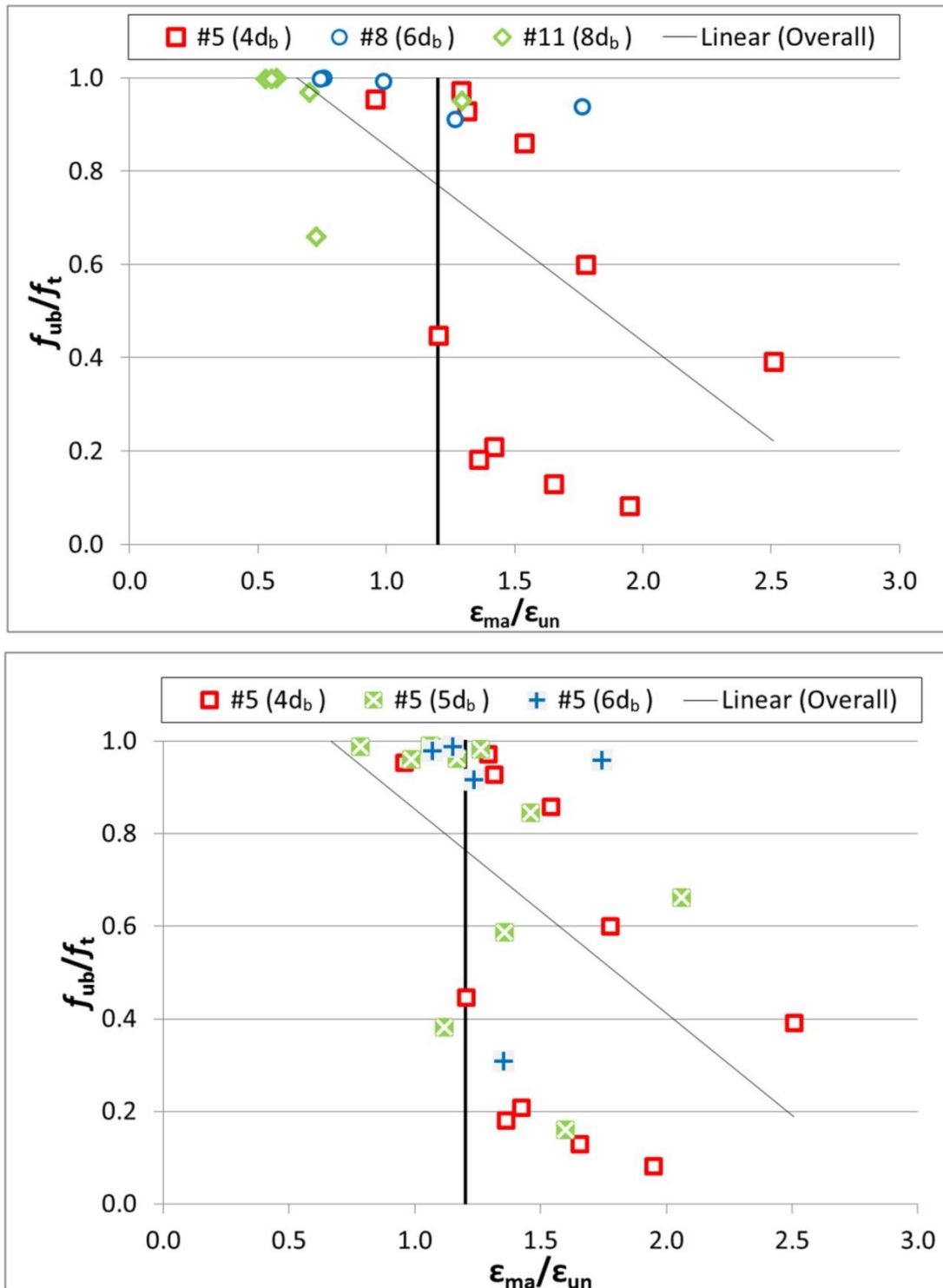


Figure 38: Normalized fracture stress (f_{ub}/f_t) vs normalized theoretical bend strain

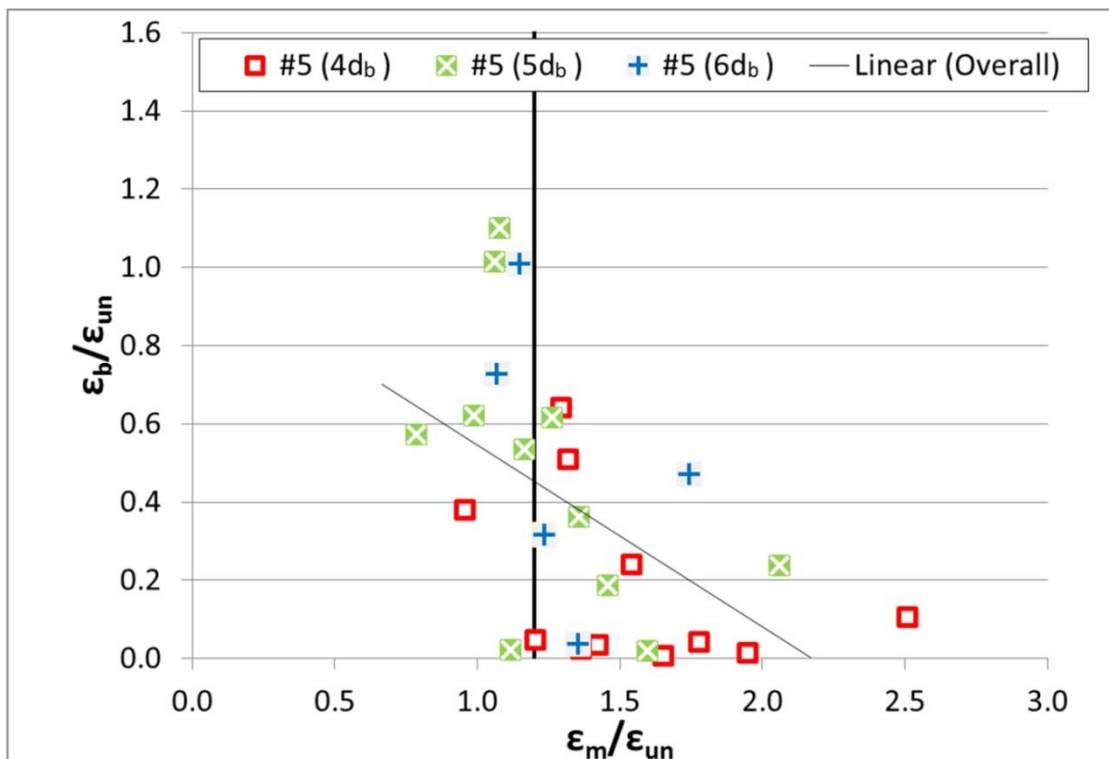
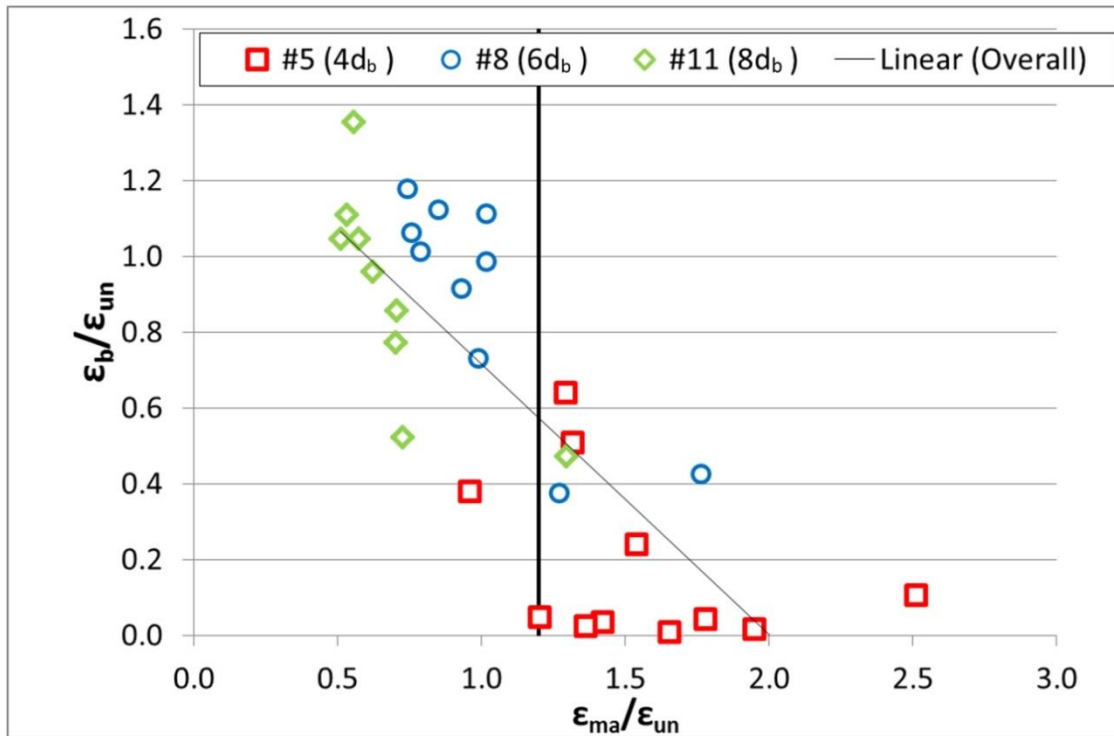


Figure 39: Normalized fracture strain vs normalized theoretical bend strain

5. Summary and Conclusions

Summary

Bend/re-bend tests were conducted on reinforcing bars with yield strength ranging from 60 ksi to approximately 120 ksi. Strain aging tests were also conducted on the bars to ensure that the bar bends were re-bent after most of the strain aging embrittlement effects had occurred. The bend/re-bend test variables were: bar grade, bar manufacturing process, bar diameter (d_b), and bend inside diameter. High-strength reinforcing bars (grade 80 and above) produced using the most prevalent methods in the U.S. were obtained from four of the main manufacturers in the U.S. Bar sizes were #5, #8, and #11 and the bars were bent to meet the minimum specified ACI 318-14 bend diameters for each of the sizes. #5 bars were bent with inside bend diameters of 4 bar diameters (transverse steel requirement) to 6 bar diameters (longitudinal steel requirements). The bar specimens were bent into a V-shape and pulled in tension until fracture. All bars were bent and tested at a temperature of about 20°C. An optical measurement system was used to record bars strains and changes in the bend angle during re-bending. Performance measures used to quantify the performance of bends included:

- a) The remaining bend angle at fracture during re-bending (θ_b)
- b) The axial strain at fracture normalized by the bar uniform elongation strain ($\epsilon_b / \epsilon_{un}$)
- c) The axial stress at fracture normalized by the bar yield strength (f_{ub} / f_y)
- d) The axial stress in bar at fracture normalized by the bar tensile strength (f_{ub} / f_t)

Conclusions

The chemistry of the bars was found to affect the extent of strain aging significantly. The higher the concentration of Vanadium in the steel, which was used by some manufacturers to increase strength, the lower the embrittlement due to strain aging was found. Overall, however, strain aging embrittlement never resulted in more than a 20% reduction in bar fracture strains.

Overall, for all bar sizes and types, as bar strength (or grade) increased, bend performance decreased as demonstrated by lower stresses, strains, and changes in the bend angle at fracture during re-bending. Moreover, for all bar sizes and types, as the bend inside diameter increased, bend performance was seen to improve.

Bends in #8 and #11 bars were found to perform adequately at the current ACI 318-14 minimum bend diameters for all grades. Most #8 and #11 bar bends strengthened fully, prior to fracturing at stresses above yield and at relatively large inelastic strains.

Bends in #5 bars showed significantly varied performance. Grade 60 #5 bars, bent to achieve a target inside diameter of $4d_b$, were able to reach stresses close to yield prior to fracture during re-bending. Bends in grade 80 and 100 bars, however, only reached fractions of their yield strength during re-bending when bent to achieve a $4d_b$ inside diameter. The performance of bends in higher grade #5 bars reached larger stresses and strains as the bend diameter was increased, with grade 80 bars reaching stresses close to their yield with $5d_b$ bends and grade 100 bars reaching yield strengths with bend diameters of $6d_b$.

The maximum strains incurred during bending were estimated and normalized by the bar uniform strain capacity ($\epsilon_{ma}/\epsilon_{un}$). The parameter was found to be negatively correlated with all performance measures. As bending strains increased with respect uniform strain capacities, bent specimens were found to sustain lower stress and strain at fracture during re-bending. To achieve specimens that reached at least their yield stress at fracture, it was found that the $\epsilon_{ma}/\epsilon_{un}$ should not exceed 1.2, which

corresponds to target inside bend diameters of no less than $4d_b$ for grade 60, $5d_b$ for grade 80, and $6d_b$ for grade 100 bars.

Currently, ACI 318-14 provides minimum bar-bend diameters as a function of bar size mainly and bar application (transverse or longitudinal bars). However, as higher and less ductile grades of bars are introduced, it may become necessary to set minimum bend diameters as a function of uniform or fracture elongations, which may reflect different performance objectives for different bar applications or demands.

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